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# Lanthanide-doped LiYF<sub>4</sub> nanoparticles: Synthesis and multicolor upconversion tuning

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### 1. Introduction

# Over the past decade, lanthanide-doped nanoparticles have left their mark as efficient luminescent nanomaterials [1]. In contrast to conventional luminescent materials, such as fluorescent organic dyes and semiconducting nanoparticles, lanthanide-doped nanoparticles generally offer narrow emission bands, long luminescence lifetimes (micro- to millisecond range), low toxicity, as well as high

resistance to photobleaching, blinking, and photochemical degradation [2]. In particular, the lanthanide-doped nanoparticles comprising proper host-dopant combinations can convert near infrared long-wavelength excitation radiation into shorter-wavelength visible emissions [3]. These unique anti-Stokes emitters, now widely known as upconversion nanoparticles, have evolved as a rapidly growing field and opened up the opportunity for creating

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#### ABSTRACT

We report the synthesis of tetragonal-phase LiYF<sub>4</sub> nanoparticles doped with upconverting lanthanide ions. The nanoparticles have been characterized by XRD, TEM, and luminescence decay studies. The size of the as-synthesized LiYF<sub>4</sub> nanoparticles can be tuned by varying the precursor ratio of F to lanthanide ions. Passivated by oleic acid ligands, the LiYF<sub>4</sub> nanoparticles can be readily dispersed in a wide range of nonpolar solvents including hexane, cyclohexane, dichloromethane, and toluene. The lanthanidedoped (Yb<sup>3+</sup>, Er<sup>3+</sup>, Tm<sup>3+</sup>, Ho<sup>3+</sup>) LiYF<sub>4</sub> nanoparticles show intense upconversion emissions upon near infrared excitation at 980 nm. By varying composition and concentration of the dopant ions, the color output can be precisely modulated under single wavelength excitation with a diode laser.

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new applications in diverse fields such as displays, solar cells, and biological assays [4].

Upconversion nanoparticles are typically composed of an inorganic host lattice and lanthanide dopant ions embedded in the host lattice [3]. Although the upconversion process primarily makes use of the ladder-like arranged 4f energy levels of the lanthanide dopant ions, the host lattice can significantly alter the upconversion processes by exerting a crystal field around the dopant ion and subtracting excitation energy of the dopant ion through lattice vibration. Therefore, a selection of appropriate host materials is essential in the synthesis of lanthanide-doped nanoparticles with favorable optical properties such as high upconversion efficiency and controllable emission profile. Upconversion emissions have been observed for a wide variety of host materials including oxides, bromides, chlorides, and fluorides [5]. Fluorides are commonly studied due to their low phonon energy and relatively high chemical stability [6]. For example, NaYF<sub>4</sub> has been generally regarded as the most efficient upconversion host for practical applications. To

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this regard, considerable efforts have been devoted to the synthesis and multicolor tuning of lanthanide-doped NaYF<sub>4</sub> upconversion nanoparticles [7].

As one of the most efficient host materials for upconversion process, LiYF<sub>4</sub> has also been frequently employed to fabricate lanthanide-doped upconversion crystals [8]. In contrast to NaYF<sub>4</sub> host matrix, LiYF<sub>4</sub> not only offers comparable emission intensity, but also generates additional emission lines [9]. Thus, LiYF<sub>4</sub> can provide us with a complementary host matrix to NaYF<sub>4</sub> for technical applications such as scintillation and tunable upconversion lasers [9b,10]. Despite the phenomenal growth in nanoparticle synthesis, there are relatively few reports on the synthesis of LiYF<sub>4</sub> nanoparticles due to formidable synthetic challenges. To the best of our knowledge, most LiYF4 nanoparticles known to date have been synthesized by thermal decomposition [9]. Although this method enables the synthesis of monodisperse nanoparticles with high dispersability in organic solvents, the requirement of toxic rare-earth trifluoroacetate precursors and release of toxic byproducts raises issues of substantial environmental concern, thereby limiting its use for commercial purposes.

In this paper, we present a facile and relatively environmental-benign approach for the synthesis of monodisperse lanthanide-doped LiYF<sub>4</sub> nanoparticles. The synthetic strategy is based on the interaction of metal oleate with ammonium fluoride at room temperature and subsequent growth of the nanoparticles at elevated temperatures. We also show that multicolor tuning of the upconversion emission can be obtained by varying doping composition and concentration, while the size of the nanoparticles could be tuned by controlling the ratio of the fluoride to metal ions.

## 2. Experimental section

#### 2.1. Materials

YCl<sub>3</sub>·6H<sub>2</sub>O (99.99%), YbCl<sub>3</sub>·6H<sub>2</sub>O (99.99%), ErCl<sub>3</sub>·6H<sub>2</sub>O (99.9%), TmCl<sub>3</sub>·6H<sub>2</sub>O (99.9%), HoCl<sub>3</sub>·6H<sub>2</sub>O (99.99%), LiOH (98%), NH<sub>4</sub>F (98%), 1-octadecene (90%) and oleic acid (90%) were purchased from Sigma-Aldrich and used as starting materials without further purification.

#### 2.2. Nanoparticle synthesis

In a typical experiment, 2 ml solutions of RECl<sub>3</sub> (0.2 M, RE = Y, Yb, Er, Ho and Tm) in methanol were added to a 50ml flask containing 3 ml of oleic acid and 7 ml of 1octadecene. The solution was heated to 150 °C for 30 min and then cooled to room temperature. Subsequently, 5 ml methanolic solution of NH<sub>4</sub>F (1.6 mmol) and LiOH (1 mmol) was added to the flask and the resulting mixture was stirred for 30 min. After removal of the methanol by evaporation, the solution was heated to 290 °C in an argon atmosphere for 1.5 h and cooled to room temperature. The resulting nanoparticles with a narrow size distribution (about 21 nm) were precipitated by addition of ethanol, collected by centrifugation, washed with water and ethanol for several times, and redispersed in cyclohexane.

#### 2.3. Characterizations

X-ray diffraction (XRD) analysis was carried out on a Siemens D5005 X-ray diffractometer with Cu Kα radiation  $(\lambda = 1.5406 \text{ Å})$ . Transmission electron microscopic (TEM) measurements were carried out using a JEOL 2010 transmission electron microscope operating at an acceleration voltage of 200 kV. The luminescence spectra were obtained with a DM150i monochromator equipped with a R928 photon counting photomultiplier tube (PMT), in conjunction with a 980 nm diode laser. The decay curves of the LiYF<sub>4</sub> were measured with a customized UV to mid infrared phosphorescence lifetime spectrometer (FSP920-C, Edinburgh) equipped with a digital oscilloscope (TDS3052B, Tektronix) and a tunable mid-band OPO laser as an excitation source (410-2400 nm, Vibrant 355II, OPOTEK). All luminescence studies were carried out at room temperature.

#### 3. Results and discussion

The size and morphology of the lanthanide-doped  $LiYF_4$ nanoparticles were first studied by transmission electron microscopy (TEM). Fig. 1a shows a representative TEM micrograph of Yb/Er (18/2 mol %) co-doped nanoparticles with an average diameter of 21 nm. High-resolution TEM (HRTEM) image reveals the highly crystalline nature of the as-synthesized LiYF<sub>4</sub> nanoparticles (Fig. 1b). The HRTEM image also shows lattice fringes with observed *d*-spacings of 0.46 nm and 0.30 nm that respectively correspond to lattice spacings in the (011) and (103) planes of tetragonalphase LiYF<sub>4</sub> (Fig. 1b). The electron diffraction pattern shown in Fig. 1c obtained from Fourier transform of the HRTEM image confirms the single-crystalline tetragonalphase of the nanoparticle. Compositional analysis of an individual particle by energy-dispersive X-ray spectroscopy (EDX) reveals the presence of the doped elemental Yb and Er (Fig. 1d). The samples were further studied by XRD and shown in Fig. 1e. All diffraction peaks can be well indexed in accordance with tetragonal LiYF<sub>4</sub> crystal (JCPDS file No. 81-2254), indicating the formation of pure tetragonal-phase of the nanoparticles. Importantly, these nanoparticles are well dispersed in a variety of nonpolar solvents including hexane, cyclohexane, dichloromethane, and toluene.

The same synthetic procedure was further used to synthesize LiYF4 nanoparticles doped with other lanthanide ions such as Yb/Tm (20/0.2 mol %) and Yb/Ho (18/ 2 mol %). TEM images in Fig. 2 a and b show that the size and morphology of the nanoparticles closely resemble to those of Yb/Er co-doped LiYF<sub>4</sub> nanoparticle counterpart. These results demonstrate that the variation of dopant ions in low concentrations does not alter the particle growth process. However, under identical experimental conditions, the size of the nanoparticles is found to be dependent on the ratio of fluoride  $(F^-)$  to rare-earth metal  $(RE^{3+})$  ions. As shown in Fig. 2c and 2d, large nanoparticles of about 40 nm were obtained as the ratio of  $F^-$  to  $RE^{3+}$  reached 8/3. With increase in the F<sup>-</sup> to RE<sup>3+</sup> ratio, a decrease in particle size was observed. Ultra small nanoparticles of about 12 nm can be obtained when the  $F^-$  to  $RE^{3+}$  ratio reached 6.



**Fig. 1.** (a) Low-resolution TEM image of the as-synthesized LiYF<sub>4</sub>:Yb/Er (18/2 mol %) nanoparticles. (b) High-resolution TEM image of a nanoparticle taken in [311] incidence and (c) the corresponding Fourier-Transform diffractogram. (d) EDX analysis and (e) XRD pattern of the nanoparticles. Note that the Li ion is undetectable by EDX and the strong signals for Cu in (d) come from the copper TEM grid.



**Fig. 2.** TEM images of the as-synthesized nanoparticles with different doping compositions and metal-to-F ratios. (a) LiYF<sub>4</sub>: Yb/Ho (18/2 mol %). (b) LiYF<sub>4</sub>: Yb/Tm (20/0.2 mol %). (c) LiYF<sub>4</sub>: Yb/Er (18/2 mol %) with a metal-to-F ratio of 3/8. (d) LiYF<sub>4</sub>: Yb/Er with a metal-to-F ratio of 1/6.



Fig. 3. Room-temperature upconversion emission spectra of LiYF<sub>4</sub> nanoparticles co-doped with (a) Yb/Er (18/2 mol %), (b) Yb/Tm (20/0.2 mol %), and (c) Yb/Ho (18/2 mol %). The samples were excited with a 980-nm diode laser operating at 600 mW. Inserts are digital camera photographs of corresponding solution samples.

The particle size variation as a function of F to rare-earth metal ratio can be explained by a crystallization speedcontrolled model [11]. Because the particle growth was initiated with the reaction between the  $F^-$  ion and rareearth oleate complex, a high concentration of  $F^-$  can significantly enhance particle crystallization rate. The crystallization acceleration effect related to high monomer concentration eventually leads to decreased critical size of the product [12].

Each lanthanide ion possesses a distinct set of energy levels that result in characteristic emissions at particular wavelengths. The upconversion emission from these nanoparticles can be readily manipulated by modifying doping compositions. Under excitation at 980 nm, the LiYF<sub>4</sub>:Yb/Er (18/2 mol %) nanoparticles exhibit characteristic sharp emission peaks resulting from  ${}^{2}H_{9/2} \rightarrow {}^{4}I_{15/2}$  (410 nm),  ${}^{2}H_{11/2}$ , ${}^{4}S_{3/2} \rightarrow {}^{4}I_{15/2}$  (525–540 nm), and  ${}^{4}F_{9/2} \rightarrow {}^{4}I_{15/2}$  (660 nm) transitions of Er<sup>3+</sup> (Fig. 3a). These emissions lead to an overall green color light (Fig. 3a, insert). In contrast, the LiYF<sub>4</sub>:Yb/Tm (20/0.2 mol %) nanoparticles exhibit a blue color emission resulting from  ${}^{1}D_2 \rightarrow {}^{3}F_4$  (450 nm),  ${}^{1}G_4 \rightarrow {}^{3}H_6$  (475 nm), and  ${}^{1}G_4 \rightarrow {}^{3}F_4$  (650 nm) transitions of Tm<sup>3+</sup> while the LiYF<sub>4</sub>: Yb/Ho (18/2 mol %) nanoparticles display a yellow color emission due to  ${}^{5}F_4 \rightarrow {}^{5}I_8$  (525 nm),  ${}^{5}F_5 \rightarrow {}^{5}I_8$  (650 nm) transitions of Ho<sup>3+</sup> (Fig. 3 b and c).

It should be noted that the Yb<sup>3+</sup> in the host lattice does not emit visible light. However, the Yb<sup>3+</sup> makes a significant contribution to the upconversion emissions by acting as a sensitizer that strongly absorbs the excitation irradiation and then transfers the excitation energy to the emitters (Fig. 4a). The energy transfer process finds reflection as a rising edge in the luminescence decay curves (Fig. 4 b–d). The decay curves also showed a nearly single-exponential decay in the tail. By fitting with a single exponential function  $I(t) = I_0 \exp(-t/\tau)$ , the lifetimes for



**Fig. 4.** (a) Schematic energy level diagrams showing typical UC processes for Yb<sup>3+</sup>, Er<sup>3+</sup>, Tm<sup>3+</sup> and Ho<sup>3+</sup>. The dashed-dotted, dotted, and full arrows represent excitation, multiphonon relaxation, and emission processes, respectively. (b–d) Luminescence decay curves of Er<sup>3+</sup> in LiYF<sub>4</sub>: Yb/Er (18/2 mol %) nanoparticles by monitoring the upconversion emission centered at 522 nm, 551 nm, and 669 nm. Flexible lines: experimental data; straight lines: fitting results by  $I(t) = I_0 \exp(-t/\tau)$  ( $I_0$  is the initial emission intensity at t = 0, and  $\tau$  is the lifetime of the emission center).

 ${}^{2}\text{H}_{11/2}$ ,  ${}^{4}\text{S}_{3/2}$ ,  ${}^{4}\text{F}_{9/2} \rightarrow {}^{4}\text{I}_{15/2}$  transitions of  $\text{Er}^{3+}$  are determined to be around 51, 54, and 56 µs, respectively. As expected, the luminescence lifetimes of  ${}^{2}\text{H}_{11/2}$  and  ${}^{4}\text{S}_{3/2}$  are the same since their populations are thermally coupled. The observed lifetimes for  ${}^{2}\text{H}_{11/2}$ ,  ${}^{4}\text{S}_{3/2} \rightarrow {}^{4}\text{I}_{15/2}$  transitions are basically consistent with those in the NaYF4 counterparts, whereas much shorter lifetime for  ${}^{4}\text{F}_{9/2} \rightarrow {}^{4}\text{I}_{15/2}$  transition is observed relative to that in NaYF4 ( $\sim 141 \text{ µs}$ ) [13]. The significant difference in lifetime for  ${}^{4}\text{F}_{9/2} \rightarrow {}^{4}\text{I}_{15/2}$  transition may be ascribed to dissimilar crystal-field surroundings of  $\text{Er}^{3+}$  ions embedded in LiYF4 and NaYF4.

In a further set of experiments, we demonstrated that upconversion color tuning can be achieved by utilizing a three-component doping system in a dual emission process. Control of the doping concentration in the host lattice enables precise modulation of the relative intensities of upconversion emissions and thus the color output. For example, the LiYF<sub>4</sub> nanoparticles triply-doped with Yb/ Er/Tm show dual emissions of  $\rm Er^{3+}$  and  $\rm Tm^{3+}$ . The relative intensity ratio of  $\rm Er^{3+}$  and  $\rm Tm^{3+}$  emissions can be manipulated by varying the doping concentration of  $\rm Er^{3+}$  in the host lattice. In the absence of  $\rm Er^{3+}$  dopant, the Yb/Tm

co-doped nanoparticle system exhibits a blue color emission as previously discussed. Upon addition of a second emitter of  $\text{Er}^{3+}$  with increased dopant concentrations (0–1 mol %), the nanoparticle system exhibits significant changes in the green ( ${}^{2}\text{H}_{11/2}$ ,  ${}^{4}\text{S}_{3/2} \rightarrow {}^{4}\text{I}_{15/2}$ ) and red ( ${}^{4}\text{F}_{9/2} \rightarrow {}^{4}\text{I}_{15/2}$ ) spectral region (Fig. 5a), resulting in tunable color output from blue to green (Fig. 5c–g).

Dual emission can also be achieved by simultaneously doping Tm and Ho in the host lattice. In comparison with Yb/Tm co-doped sample, the Yb/Tm/Ho triply-doped system exhibits additional emission peaks of Ho in the green and red spectral region (Fig. 5b). This dual emission process provides an alternative route towards upconversion multicolor tuning. As the doping concentration of Ho<sup>3+</sup> increases, the colloidal solution of the triply-doped nanoparticles shows a notable color change (Fig. 5 h and i). The absence of quenching in the emission intensity as evidenced by the examination through the naked eye in all the triply-doped samples indicates that there is negligible energy transfer and cross relaxation between different types of activators. This phenomenon can be attributed to the low concentration (< 1 mol) of dopant emitter ions in the host lattice [14].



**Fig. 5.** Room temperature upconversion emission spectra of (a) LiYF<sub>4</sub>: Yb/Tm/Er (20/0.2/0–1 mol %) and (b) LiYF<sub>4</sub>: Yb/Tm/Ho (20/0.2/0–0.5 mol %) nanoparticles dispersed in cyclohexane solutions (8 mM). The spectra in (a) and (b) were normalized to  $Tm^{3+}$  480 nm emission. (c–i) Compiled luminescent photos showing corresponding colloidal solutions of LiYF<sub>4</sub>:Yb/Tm/Er (20/0.2/0–1 mol %) and LiYF<sub>4</sub>:Yb/Tm/Ho (20/0.2/0–0.5 mol %), respectively.

### 4. Conclusions

In this work, tetragonal-phase LiYF<sub>4</sub> nanoparticles doped with upconverting lanthanide ions were synthesized by a novel solution-based method. Monodisperse nanoparticles with a tunable size can be readily obtained by varying the ratio of fluoride to rare-earth metal ions. The as-synthesized nanoparticles show intense upconversion emissions upon excitation at 980 nm.

The color output of the upconversion emission can be precisely manipulated by varying doping composition and concentration, providing potential applications for multiplex biological labeling and imaging.

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