

Pressure-Induced Phase Transformation and Band-Gap Engineering of Formamidinium Lead Iodide Perovskite Nanocrystals

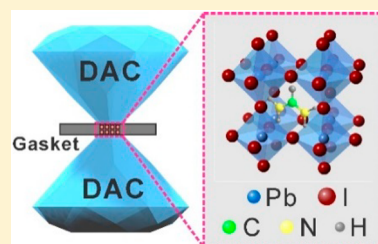
Hua Zhu,[†] Tong Cai,[†] Meidan Que,[†] Jeong-Pil Song,[†] Brenda M. Rubenstein,[†] Zhongwu Wang,[‡] and Ou Chen^{*,†}

[†]Department of Chemistry, Brown University, Providence, Rhode Island 02912, United States

[‡]Cornell High Energy Synchrotron Source, Cornell University, Ithaca, New York 14853, United States

Supporting Information

ABSTRACT: Formamidinium lead halide (FAPbX₃, X = Cl, Br, I) perovskite materials have recently drawn an increased amount of attention owing to their superior optoelectronic properties and enhanced material stability as compared with their methylammonium-based (MA-based) analogues. Herein, we report a study of the pressure-induced structural and optical evolutions of FAPbI₃ hybrid organic–inorganic perovskite nanocrystals (NCs) using a synchrotron-based X-ray scattering technique coupled to in situ absorption and photoluminescence spectroscopies. As a result of their unique structural stability and soft nature, FAPbI₃ NCs exhibit a wide range of band-gap tunability (1.44–2.17 eV) as a function of pressure (0–13.4 GPa). The study presented here not only provides an efficient and chemically orthogonal means to controllably engineer the band gap of FAPbI₃ NCs using pressure but more importantly sheds light on how to strategically design the band gaps of FA-based hybrid organic–inorganic perovskites for various optoelectronic applications.



Hybrid organic–inorganic perovskites consisting of an organic cation caged within an inorganic cuboctahedral framework^{1–4} have recently drawn a significant amount of attention owing to their unique crystal structure, high defect tolerances, and superior optoelectronic properties.^{2,4–17} To this extent, formamidinium lead iodide (FAPbI₃) perovskites stand as one of the most promising candidates for optoelectronic applications among all the perovskite materials owing to their small band gap (i.e., ~1.45 eV), balanced ionic sizes and thus high thermal and moisture stabilities, long photocarrier lifetime and diffusion length, and large optical absorption coefficients.^{7,18,19} Altogether, these properties strongly suggest that FAPbI₃ perovskites have the potential to become cornerstones of all future optoelectronic devices, including solar cells.^{7,19–24} In addition, scaling these materials to the nanoscale sizes of colloidal FAPbI₃ nanocrystals (NCs) improves their properties over their bulk counterparts: NCs can be processed in solution, their band gaps may be tuned by changing particle size and shape, and NCs exhibit enhanced phase stability and mixability with other functional materials.^{8,9,25} Hence the study of the structural and optical behaviors, as well as their correlations, of FAPbI₃ hybrid perovskite NCs is not only fundamentally important but also technologically relevant. To this end, diamond anvil cell (DAC)-based high-pressure techniques in conjunction with in situ structural and optical characterizations have recently arisen as a fast and convenient means to study, process, modify, and improve nanomaterials.^{26–36} To date, various perovskite systems have been investigated using this technique at extreme pressures.^{36–44} Interesting structure–property relationship diagrams of perovskite materials have been reported, including

those of bulk FA-based hybrid perovskites.^{37,45,46} However, such studies have yet to be extended to understanding FA-based nanoscale systems.

Herein, we present a pressure-processing study of high-quality FAPbI₃ hybrid organic–inorganic perovskite NCs across a wide range of pressures using a DAC. Simultaneous characterization of structural and optical properties allowed us to reveal not only the crystal phase transformations of the materials but also the structure–optical–property correlations in situ. In contrast with related bulk studies, our study shows that the FAPbI₃ NCs undergo a phase transition from a perfect cubic lattice of *Pm* $\bar{3}m$ to a cubic supercell of *Im* $\bar{3}$ without assuming the commonly observed orthorhombic (*Pnma*) phase before amorphization. This high structural stability combined with the low bulk modulus of FAPbI₃ perovskite NCs leads to the widest range of band-gap tunability of all other studied perovskite materials. Our study sheds light onto how to design band gaps of FA-based perovskite NC materials for a wide spectrum of applications.

FAPbI₃ hybrid perovskite NCs were synthesized using a previously published method (see the [Experimental Section](#) and the [Supporting Information \(SI\)](#)).⁹ Figure 1A shows the absorption and photoluminescence (PL) spectra of the FAPbI₃ NCs. Transmission electron microscopy (TEM) measurements show that the FAPbI₃ NCs have a cubic shape with an average edge length of 19.0 ± 3.4 nm (Figure 1B and Figure S1). The high-resolution TEM (HR-TEM) measurements

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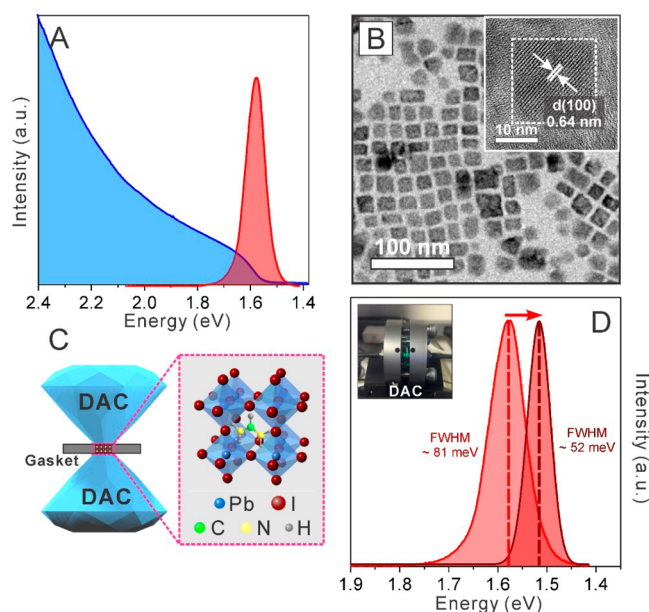


Figure 1. (A) Absorption (blue) and PL (red) spectra of the FAPbI₃ perovskite NCs in toluene. (B) Typical TEM image of the FAPbI₃ perovskite NCs as a cubic shape with an average edge length of 19.0 ± 3.4 nm. Inset shows a high-resolution (HR)-TEM image of one particle. (C) Schematic of FAPbI₃ NCs inside a DAC. (D) PL spectra of the NCs in solution (red line) and in the DAC (dark red line). Inset shows a DAC loaded with NCs.

clearly reveal the atomic lattice fringes with a d spacing of 6.4 Å, corresponding to the (100) plane of the FAPbI₃ cubic crystal phase (space group: $Pm\bar{3}m$).³ The as-synthesized FAPbI₃ NC–toluene suspension was slowly dried and loaded into a DAC. The PL peak was red-shifted ~ 71 meV (~ 37 nm) and became $\sim 35\%$ narrower (81 vs 52 meV) than that of the solvent-dispersed NCs, indicating that the reabsorption and re-emission processes occur within a densely packed NC film.⁴⁷

To characterize the crystal phase evolution of FAPbI₃ NCs under pressure, HR synchrotron-based wide-angle X-ray scattering (WAXS) patterns were collected upon applying pressure. Figure 2 shows a series of WAXS patterns collected during a pressure cycle of 0–13.4 GPa. At ambient pressure, FAPbI₃ NCs had a perfect cubic crystal phase (space group: $Pm\bar{3}m$) with a lattice constant of $a = 6.351$ Å (Figure 2H, Figure S2, and Table S2), in good agreement with HR-TEM measurements (Figure 1B, inset). Upon compression, the cubic phase remained stable up to ~ 0.6 GPa, whereas all of the WAXS peaks shifted to larger angles, indicating a uniform shrinkage of the atomic lattice. When the pressure reached 0.6 GPa, two additional scattering peaks emerged at q values of 15.9 and 18.8 nm^{−1} (Figure 2E and Figure S4), which were well assigned to the (310) and (321) planes of the $Im\bar{3}$ crystal phase of FAPbI₃, respectively. This observation suggested that the FAPbI₃ NCs underwent a phase transition from $Pm\bar{3}m$ to $Im\bar{3}$, in line with previous reports on FA-based lead halide bulk crystal materials.⁴⁵ The emergent tilting of [PbI₆]^{4−} octahedra resulted in an enlarged ($2 \times 2 \times 2$) supercell lattice (Figure

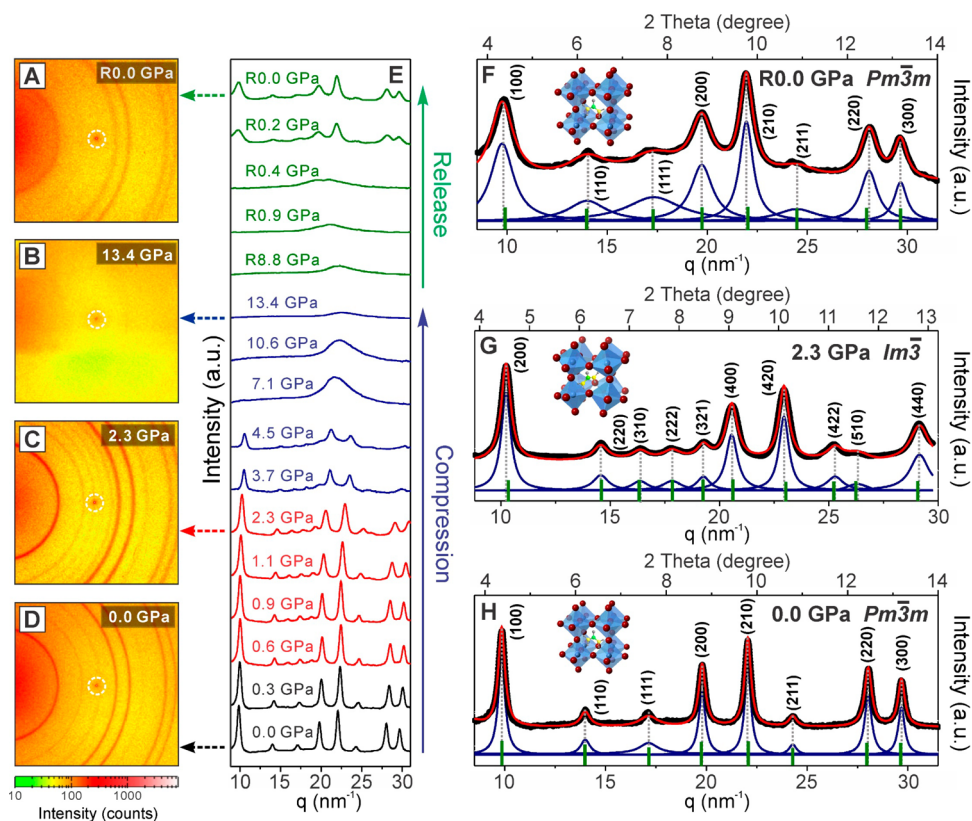


Figure 2. (A–D) Integrated 2D WAXS patterns of FAPbI₃ NCs during a sequential compression–decompression cycle. The white dashed circles are noise signals on the detector. The signal intensity is shown in the scale bar below. (E) WAXS patterns at different pressures during the compression and decompression processes. (F–H) Integrated WAXS spectra with calculated Bragg reflection positions at R0.0 (after decompression), 2.3, and 0.0 GPa, respectively. The green bars show the calculated Bragg reflection positions of each phase. At 0.0 and R0.0 GPa, FAPbI₃ NCs show a perfect cubic $Pm\bar{3}m$ crystal phase. At 2.3 GPa, NCs show a cubic supercell $Im\bar{3}$ crystal phase.

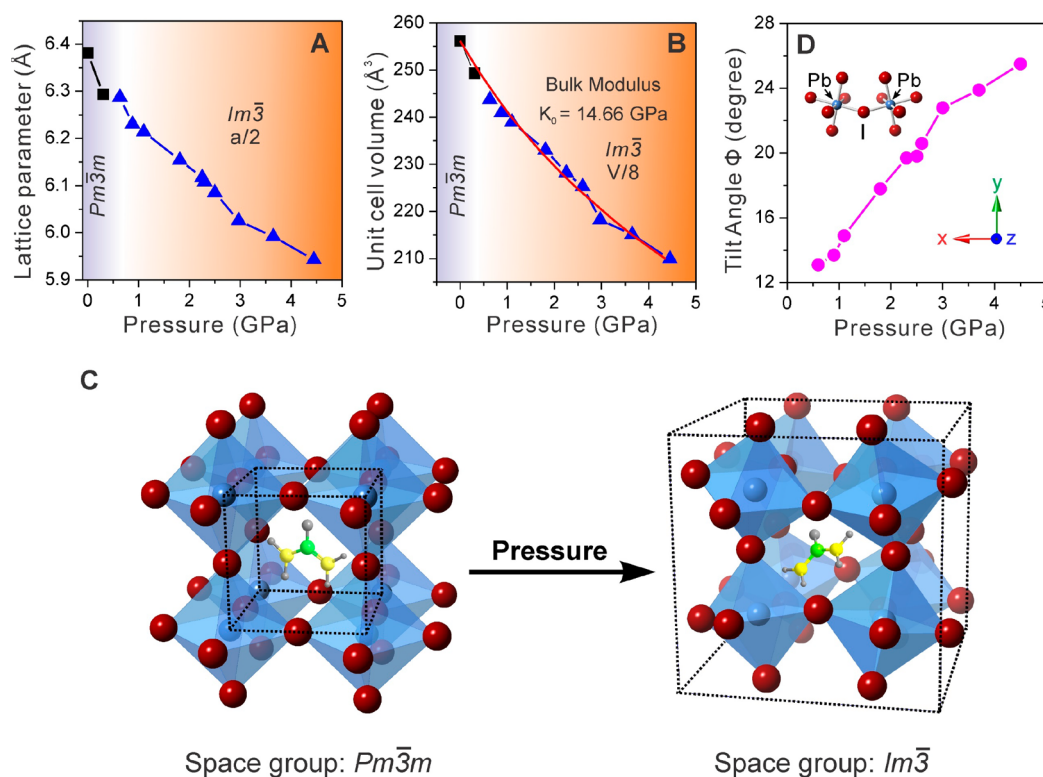


Figure 3. Plots of the evolution of the FAPbI₃ structure as a function of pressure. (A) Lattice parameter evolution during pressure. Lattice parameters are normalized (a for $Pm\bar{3}m$ crystal phase (square) and $a/2$ for $Im\bar{3}$ crystal phase (triangle)) for a fair comparison. (B) Evolution of normalized unit-cell volumes. The red line is a fitted curve using the second-order Birch–Murnaghan equation of state, giving a bulk modulus of 14.66 GPa. (C) Unit-cell schematics of FAPbI₃ perovskite NCs with $Pm\bar{3}m$ (left panel) and $Im\bar{3}$ (right panel) crystal structures. The $[PbI_6]^{4-}$ octahedron was tilted along the $[111]$ cubic direction in the $Im\bar{3}$ phase. Black dotted lines indicate the unit-cells. (D) Octahedral tilt angle evolution as a function of pressure.

S3) as compared with the initial FAPbI₃ cubic phase ($Pm\bar{3}m$).^{4,45} Upon further compression, all of the scattering peaks red-shifted while maintaining the crystallographic relationship, indicating a preservation of the $Im\bar{3}$ phase (Figure 2E). In contrast with the MAPbI₃ NCs and FAPbBr₃ bulk crystal cases, no subsequent phase transition to the highly distorted orthorhombic phase (space group: $Pnma$) was observed over the pressure range of 2–3 GPa in this study (Figures S4–S7 and Tables S3–S6). The WAXS pattern at 2.3 GPa is shown in Figure 2C as one example. All of the scattering peaks are highly symmetric and do not show any noticeable splitting (Figure 2G). The pattern can be refined into a perfect cubic phase ($Im\bar{3}$), giving rise to a measured supercell lattice constant of $a = 12.22 \text{ \AA}$ (i.e., $a/2 = 6.11 \text{ \AA}$) (Figure S6 and Table S5). This improved structural stability is most likely caused by four short hydrogen bonds ($NH\cdots I$) formed between the FA^+ cation and I^- anions of the inorganic perovskite sublattice within individual NCs.^{3,48–50} When the applied pressure surpassed 3 GPa, the WAXS patterns started to show a diffuse background and decreased scattering intensity with a broadening peak profile, indicating the onset of a partial amorphization process, as observed in other perovskite NC systems.^{40,45,51–53} When the pressure reached 7.1 GPa, the perovskite structure totally disappeared, resulting in complete amorphization, as indicated by a single broad scattering feature at q values of $\sim 22\text{--}25 \text{ nm}^{-1}$ in the WAXS patterns (Figure 2E). Upon release of pressure, the amorphous state was preserved until the pressure was below 0.4 GPa (Figure 2A). A complete pressure removal induced a rapid

perovskite crystallization to the original cubic phase (R0.0 GPa, $Pm\bar{3}m$, Figure 2E,F). Notably, all of the scattering peaks were broader than those in the original WAXS pattern (Figure 2F), indicating a partial loss of crystallinity of the FAPbI₃ NCs. TEM measurements of the pressed sample confirmed our hypothesis, which showed no significant changes in particle size and shape (Figures S8 and S9).

To understand pressure-induced atomic motions during the phase transitions, we conducted a detailed structural analysis of FAPbI₃ NCs before amorphization. Within the pressure range of 0–0.6 GPa, FAPbI₃ NCs showed a perfect cubic structure, which maintained the Pb–I–Pb bond angle of 180° . The shrinkage of the lattice parameter and thus the decrease in unit-cell volume was a direct consequence of Pb–I bond contraction (Figure 3A,B). The continuous increase in pressure led to a distorted perovskite structure (space group: $Im\bar{3}$) with a $[PbI_6]^{4-}$ octahedral tilt of equal magnitude about all three Cartesian axes (Glazer notation: $a^+a^+a^+$, Figure 3C and Figure S10).⁵⁴ The tilting angle (Φ) with respect to the $[111]$ crystal axis increased from 13.1° at 0.6 GPa to 25.5° at 4.5 GPa (Figure 3D and Table S7). This experimental observation was consistent with the high rigidity of the $[PbI_6]^{4-}$ octahedral units as compared with the $[FAI_{12}]^{11-}$ cuboctahedral ones.⁴⁵ On the basis of our structural assignment, the volumetric compression as a function of pressure gave a bulk modulus (K_0) of 14.66 GPa via fitting of the second-order Birch–Murnaghan equation of state (Figure 3B), comparable to previous reports for MAPbI₃.^{37,40} Note that this bulk modulus (K_0) of FAPbI₃ NCs is significantly lower than

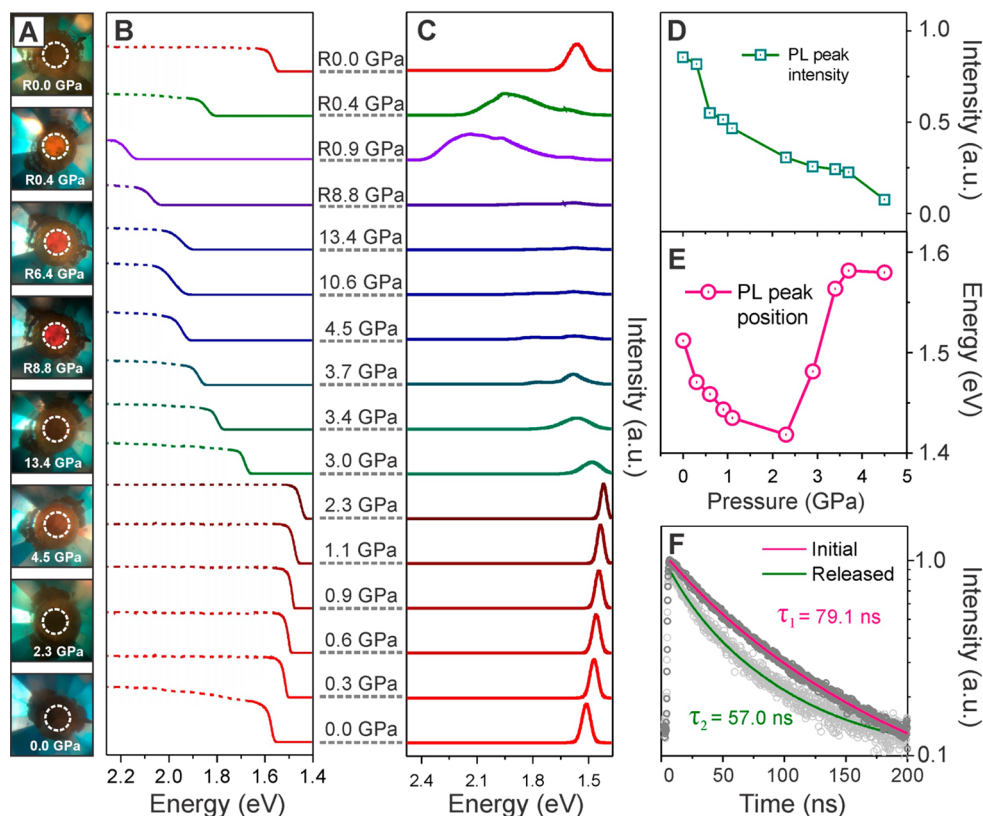


Figure 4. (A) Series of photographs of FAPbI₃ NCs in a DAC at different pressures. (B) Absorption and (C) PL evolutions during the pressure cycle. Dashed lines indicate the possible signal saturation. (D) PL peak intensity evolution before complete amorphization. The PL intensity shows a continuous decrease with increasing pressure. (E) PL peak position evolution before complete amorphization. The peak position first red-shifts, then blue-shifts, and finally remains at constant value. (F) PL lifetime decay curves of the FAPbI₃ NCs before and after pressure treatment.

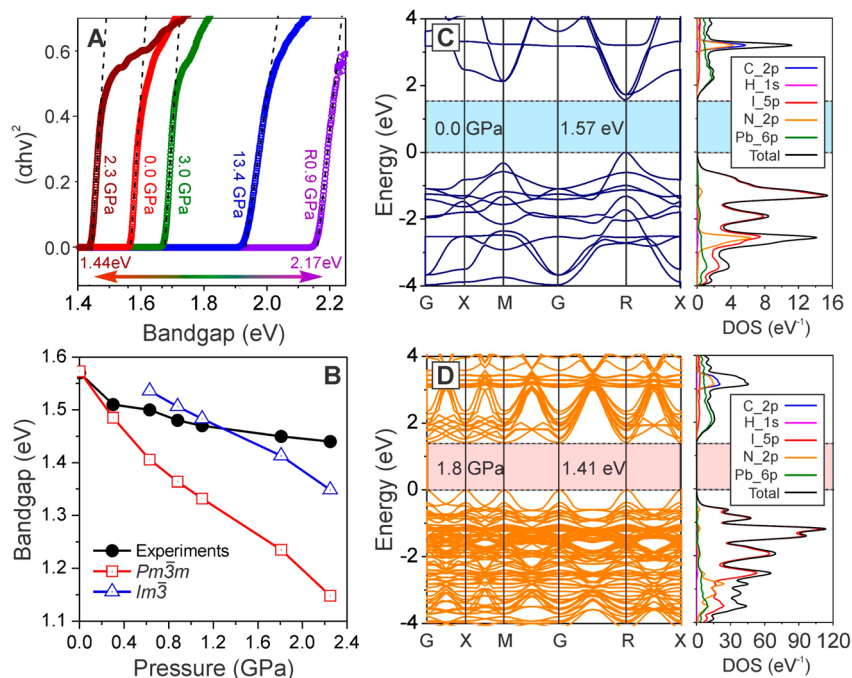


Figure 5. (A) Tauc plots of FAPbI₃ NCs at different pressures showing the absorption onset. (B) Band-gap evolution based on experimental data and DFT calculations for both the *Pm*3m and *Im*3 (corresponding to $U = 1.2$ and 0 eV) crystal phases. (C,D) Calculated band structures of FAPbI₃ NCs at 0.0 and 1.8 GPa (left panels) and the corresponding DOS (right panels), respectively.

that of all-inorganic perovskites, such as CsPbBr₃ and CsPbI₃.^{26,51} The soft nature of FAPbI₃ NCs results in a larger

distortion of the Pb–I–Pb bond and hence a smaller coupling between Pb 6p orbitals and I 5p orbitals under pressure, which,

in principle, would enable a wide range of tunability for the pressure-engineered band gap of FAPbI₃ NCs.³⁷

To understand the behavior of the optical properties of FAPbI₃ NCs under high pressure, we performed in situ absorption and PL measurements during the pressure cycle. The entire optical profile progression was strongly coincident with the pressure-induced phase transition and amorphization processes of the FAPbI₃ NCs. In detail, continuous red shifts of both absorption (from 1.57 to 1.44 eV) and PL spectra (from 1.51 to 1.42 eV) were observed in the pressure range of 0–2.3 GPa (Figure 4A–E). This can be attributed to the Pb–I bond contraction, which leads to a greater orbital wave function overlap, thus narrowing the band gap.^{37,43} Upon further compression, the absorption band and PL peak started blue-shifting, whereas the PL intensity decreased dramatically and became nearly undetectable when the pressure was increased above ~5 GPa (Figure 4C–E). The decrease in PL intensity was attributed to the pressure-induced band gap widening, which creates deep intraband trap states with fast nonradiative relaxation pathways.^{40,45} Interestingly, during the decompression process, the absorption band kept blue-shifting until the pressure was released to 0.9 GPa, where a broad PL peak at ~2.1 eV emerged (Figure 4C) while retaining the amorphous phase of NCs (Figure 2E). Although the origin of this amorphous state PL remains unclear, it is likely related to a short-range order of highly tilted [PbI₆]^{4–} octahedra inside the amorphous FAPbI₃ NCs. When the pressure was released below ~1.0 GPa, the NC surface states became less strained with fewer defects and shallower surface-related exciton trap states,^{26,55,56} resulting in an enhanced amorphous state emission, as we observed. Further releasing pressure resulted in reversible red shifts of both the absorption feature and the PL peak, and the PL peak center finally bounced back to 1.56 eV when the pressure was completely removed (Figure 4B,C). The final PL profile exhibited higher energy (~50 meV), a broader line width (~100%), and shorter-lived charge carriers (lifetime of 57.0 vs 79.1 ns; see the details in Table S1) than the initial PL of the FAPbI₃ NCs before pressurization (Figure 4C,F). These observations can be explained by the poor crystallinity, less ordered surface atoms, and ligand states of the final NCs after pressure-treatment, consistent with the WAXS data sets (Figure 2E,F). TEM characterizations of the pressed FAPbI₃ NCs did not show significant alterations in particle sizes and shapes (Figures S8 and S9), supporting our explanation.

Because band-gap engineering of perovskite materials is essential for photovoltaic and other applications, we performed a detailed band-gap analysis using Tauc plot fitting of the linear region of the absorption band spectra and density functional theory (DFT) band structure calculations as a function of applied pressure (Figure 5, Figures S11–S13).⁵⁷ Figure 5A shows the band-gap evolution determined by a Tauc plot at different pressures. Remarkably, the pressure-tuned band gap spanned a significantly wider range (~0.7 eV) than all other reported perovskite materials across different length scales (Figure 5A and Figure S12).^{37,39,41,43–45,51} Although the band gap in an amorphous phase cannot be quantitatively determined, detailed band structures were calculated using first-principles DFT calculations for the two crystal phases (i.e., *Pm3m* and *Im3*). The density of states (DOS) calculation results clearly revealed that the valence band (VB) mainly consists of I 5p orbitals, whereas the conduction band (CB) is dominated by Pb 6p orbitals (Figure 5C,D and Figure S13).

Consequently, the high-pressure-induced band-gap evolution was dictated by the structural behavior of the [PbI₆]^{4–} octahedral subunits. Below 0.6 GPa, homogeneous contraction of [PbI₆]^{4–} octahedra resulted in a continuous shrinkage of the Pb–I bond, directly promoting the two p orbitals' overlap, thus narrowing the band gap efficiently (Figure 5B and Figure S13). When the crystal phase transitioned to *Im3*, drastic [PbI₆]^{4–} octahedral tilt resulted in reducing the Pb–I–Pb bond angle from 180°. This off-aligned Pb–I–Pb bond diminished the symmetry match between the Pb 6p orbital and the I 5p orbital. As a result, a slower band-gap narrowing process was observed from both the Tauc plot analysis and DFT calculations (Figure 5B). Above 3.0 GPa, the subsequent structural amorphization of FAPbI₃ NCs induced a large band-gap increase. Although the physical and thus the electronic structures of the amorphous state of FAPbI₃ NCs cannot be precisely determined, significant [PbI₆]^{4–} octahedral tilt and crystal domain reduction would reasonably be responsible for the further band-gap widening.

In conclusion, we probed the pressure-induced structural and optical property evolutions of FAPbI₃ hybrid organic–inorganic perovskite NCs within a pressure cycle of 0–13.4 GPa. The FAPbI₃ NCs underwent a phase transition from a cubic *Pm3m* to a cubic supercell *Im3* in the pressure range of 0–3 GPa, followed by an amorphization process. Evolutions of the absorption and PL profiles of the FAPbI₃ NCs as a function of pressure show a strong correlation with the crystal structure changes, consistent with the DFT band structure calculations. Importantly, the high structural stability with a low bulk modulus of FAPbI₃ NCs leads to a wide range of band-gap tunability from 1.44 to 2.17 eV, which is significantly wider than that of all other reported perovskite materials within a similar range of pressure cycle. Such a high band-gap engineering capability is crucial in optoelectronic applications of hybrid organic–inorganic perovskite NC materials. Our study provides important design rules for generating FA-based hybrid perovskite materials with optimized structure-determined optical properties in and even beyond optoelectronic applications.

EXPERIMENTAL SECTION

FAPbI₃ NCs were synthesized using a modified literature method.⁹ Synchrotron-based WAXS measurements²⁸ and in situ absorption (Abs) and PL characterizations were performed at the B1 station of Cornell High Energy Synchrotron Source (CHESS), Cornell University. A diamond anvil cell (DAC) was used for the high-pressure experiment. In situ PL measurements were performed through the DAC using an Ocean Optics and Thorlabs setup, in which the sample was illuminated by a 365 nm diode light source and the signal was collected by Ocean Optics spectrometer (USB2000+). In situ Abs measurements were performed through the DAC using an Ocean Optics setup, in which the sample was illuminated by white-light source from Thorlabs (SLS201L) and the signal was collected by an Ocean Optics spectrometer (USB2000+). First-principles density functional theory (DFT) calculations were carried out based on the generalized gradient approximation (GGA) within DFT using the Quantum Espresso (QE) program. We employed the GGA+*U* method in accordance with the Hubbard *U* approach.^{58,59} The calculations were performed using the Perdew–Burke–Ernzerhof (PBE) exchange-correlation functional. See the SI for experimental and computational details.

■ ASSOCIATED CONTENT

● Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.jpclett.8b01852.

Additional experimental and computational details, WAXS analysis FAPbI₃ NCs under pressure, and band-gap calculations (PDF)

■ AUTHOR INFORMATION

Corresponding Author

*E-mail: ouchen@brown.edu.

ORCID

Hua Zhu: 0000-0003-2733-7837

Zhongwu Wang: 0000-0001-9742-5213

Ou Chen: 0000-0003-0551-090X

Notes

The authors declare no competing financial interest.

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