# SCIENTIFIC REPORTS

Received: 12 May 2017 Accepted: 25 October 2017 Published online: 07 November 2017

## **OPEN** Pathway Towards High-Efficiency **Eu-doped GaN Light-Emitting** Diodes

Ioannis E. Fragkos<sup>1</sup>, Chee-Keong Tan<sup>1,4</sup>, Volkmar Dierolf<sup>2</sup>, Yasufumi Fujiwara<sup>3</sup> & Nelson Tansu<sup>1</sup>

A physically intuitive current injection efficiency model for a GaN:Eu quantum well (QW) has been developed to clarify the necessary means to achieve device guantum efficiency higher than the state-ofthe-art GaN:Eu system for red light emission. The identification and analysis of limiting factors for high internal quantum efficiencies (IQE) are accomplished through the current injection efficiency model. In addition, the issue of the significantly lower IQE in the electrically-driven GaN:Eu devices in comparison to the optically-pumped GaN:Eu devices is clarified in the framework of this injection efficiency model. The improved understanding of the quantum efficiency issue through current injection efficiency model provides a pathway to address the limiting factors in electrically-driven devices. Based on our developed injection efficiency model, several experimental approaches have been suggested to address the limitations in achieving high IQE GaN:Eu QW based devices in red spectral regime.

In recent years, III-Nitride based alloys have been placed at the frontiers of semiconductor technologies. The use of III-Nitrides has found places in a wide span of technological applications including bio-applications, thermoelectrics, solar cells, power electronics, optoelectronics and photonics<sup>1-10</sup>. Among III-Nitrides, InGaN-based alloys are of great interest. The versatility to tune the band-gap of InGaN-based alloys within the UV and visible spectral regime have established them as the main technologies driving the light emitting diode (LED) innovations<sup>11-13</sup>. The possibility for monolithically integrated red-green-blue (RGB)  $In_xGa_{1-x}N$ -based LEDs, will also open a new era for smart and ultra-efficient solid stated lighting technology in the future<sup>14</sup>.

Despite the rapid development of high efficiency InGaN-based quantum well (QW) LEDs in the UV, blue and green spectral regime, the realization of InGaN QW-based LEDs with high efficiency at wavelengths towards to red spectra regime is challenging. The following issues associated with high indium content in the active region such as, phase separation of InGaN alloy and high polarizations fields, are detrimental to internal quantum efficiency (IQE) of the devices<sup>15-19</sup>. Several works in recent years have suggested innovative approaches with the potential of achieving high efficiency for InGaN based QW LED towards red spectral regime. These works include the investigation of staggered InGaN QWs, strain compensated InGaN QWs, alternative substrates and buffer layers of InGaN QWs, InGaN with AlGaN interlayer, semipolar and non-polar InGaN QW, InGaN-delta-InN QW and InGaN/dilute-As GaNAs interface QW<sup>20-29</sup>. Despite the demonstration of InGaN based LED in the red spectra regime, the highest reported external quantum efficiency (EQE) is 2.9% which is much lower from the blue and green InGaN-based QW LEDs<sup>26</sup>.

An alternative approach of achieving red emission without the need of high In-content of InGaN based alloys is the incorporation of rare earth elements into GaN (e.g Europium)<sup>30-32</sup>. The possibility of introducing Europium element (Eu) into the GaN material has enabled the realization of GaN:Eu red light emitting devices including LEDs in the past decade<sup>33-49</sup>. However, the internal quantum efficiency (IQE) of the GaN:Eu emitter is low (<1%), despite the recent years of effort in improving the device performance. These efforts include improving the GaN:Eu material quality and utilizing heterostructures for higher IQE<sup>32,44,47–49</sup>. Improving the IQE of the GaN:Eu devices is necessary for practical technological implementation. In addition, another major obstacle is found to be the IQE discrepancy between the electrically driven and optically excited GaN:Eu devices.

<sup>1</sup>Center for Photonics and Nanoelectronics, Department of Electrical and Computer Engineering, Lehigh University, Bethlehem, PA, 18015, USA. <sup>2</sup>Department of Physics, Lehigh University, Bethlehem, PA, 18015, USA. <sup>3</sup>Division of Materials and Manufacturing Science, Graduate School of Engineering, Osaka University, Suita, Osaka, 565-0871, Japan. <sup>4</sup>Department of Electrical and Computer Engineering, Clarkson University, Potsdam, NY, 13699, USA. Correspondence and requests for materials should be addressed to I.E.F. (email: iof213@lehigh.edu) or N.T. (email: tansu@lehigh.edu)



**Figure 1.** Model of the trap assisted excitation of  $Eu^{+3}$  ion in GaN:Eu QW active region. (a) The confined electron-hole in the GaN:Eu QW are captured by the traps (purple arrows) which are close to the vicinity of  $Eu^{+3}$  ion and results in (b) complex formation. (c) After the complex formation the electron-hole pair can recombine at the trap level by releasing a non-radiative energy to the crystal lattice (brown arrow) or release a non-radiative energy used for the excitation of the nearby  $Eu^{+3}$  ion (energy transfer process-gold arrow) or it can dissociate by releasing the electron-hole back to the GaN:Eu QW. Similarly, the excited  $Eu^{+3}$  can recombine non-radiative y by releasing energy to the crystal lattice (brown arrow) or release non-radiative energy for complex formation (energy back-transfer process, dark blue arrow) or recombine radiatively with photon emission (red arrow).

Interestingly, the IQE of electrically-driven GaN:Eu devices is much lower than that of the optically-pumped GaN:Eu devices. Despite the fact that optically-pumped devices exhibited an increase in the output power and consequently in the IQE over the years, the electrically-driven devices showed a saturation in the output power, probably due to the IQE limitation of the device<sup>50</sup>. This discrepancy is possibly attributed to the dependency of the IQE on the current injection efficiency of the GaN:Eu active region for the two different excitation ways. The need of electrically-driven device is however arguably stronger than optically-pumped device because in many applications including LEDs, the devices are typically driven by injected current to achieve emission. If the GaN:Eu device is to be employed for the light emitting applications, the understanding of the factors which lead to low efficiency in electrically-driven GaN:Eu device needs to be enhanced. Thus, developing a current injection efficiency model of the GaN:Eu active region will provide a qualitative picture and a better understanding of the IQE of both optically-pumped and electrically-driven GaN:Eu red light emitters. Besides, the model can further provide the opportunity to enhance the design and fabrication of high efficiency GaN:Eu based red light emitters.

This work presents the development of physically intuitive current injection efficiency model for a GaN:Eu QW active region for understanding the discrepancy between the efficiencies of optically-pumped and electrically-driven GaN:Eu QW based devices. The discrepancies between the optically-pumped and electrically-driven RE-doped GaN LEDs devices can be explained from the differences on the carrier injection processes in the two types of devices. This study identifies and explains the limiting factors for the low IQE of the GaN:Eu QW based devices.

### Excitation path of Eu<sup>+3</sup> ions in GaN:Eu QW and current injection efficiency models

**GaN:Eu OW active region considerations.** The light emission from RE-doped semiconductors arises from the transitions between the intra 4 f electronic states of the RE ion. In the case of Eu-doped GaN semiconductor, the GaN acts as the host to the  $Eu^{+3}$  ions. The excitation of  $Eu^{+3}$  ion in GaN host is known to be mediated by traps which are present in the vicinity of  $Eu^{+3}$  ion. These traps capture free electron-hole pairs from GaN host and transfer the non-radiative recombination energy to a nearby  $Eu^{+3}$  ion<sup>34,45,47,50–53</sup>.

Our analysis in this work is carried out based on the model of a trap assisted excitation path of Eu<sup>+3</sup> ion in the GaN:Eu QW active region with  $Al_xGa_{1-x}N$  barriers as shown in Fig. 1. We represent the role of traps by a single trap level but note that the extended nature of these traps close to the vicinity of Eu<sup>+3</sup> ions could also result in several levels. The electron-hole pair capture from the trap with a characteristic rate  $1/\tau_{c_ccap}$  is notated as complex (bound-exciton) in our model, as shown in Fig. 1b. The subsequent recombination of carriers at the trap level (i.e. de-excitation of complex, Fig. 1c) releases a non-radiative recombination energy that leads to the following possible reactions: (a) excitation of the nearby Eu<sup>+3</sup> ion with a characteristic energy transfer rate of  $1/\tau_{tr}$ , (b) energy to the crystal lattice with a characteristic rate of  $1/\tau_{c_cheat}$ . In addition, after the formation of complex, another process can occur which results in the electron-hole population of the QW with a characteristic rate of  $1/\tau_{diss}$  (complex dissociation process). Additional mechanisms to consider including the consequence of the then-de-excitation of Eu<sup>+3</sup> ions: a) photon release with a characteristic rate of  $1/\tau_{rad}$ , b) non-radiative de-excitation with a characteristic rate of  $1/\tau_{tu}$ , b) non-radiative de-excitation with a characteristic rate of  $1/\tau_{tu}$ , b) non-radiative de-excitation with a characteristic rate of  $1/\tau_{tu}$ , b) non-radiative de-excitation with a characteristic rate of  $1/\tau_{tu}$ , b) non-radiative de-excitation with a characteristic rate of  $1/\tau_{tu}$ , b) non-radiative de-excitation with a characteristic rate of  $1/\tau_{tu}$ , b) non-radiative de-excitation with a characteristic rate of  $1/\tau_{tu}$ . In addition, the carrier processes related to the GaN host and the  $Al_xGa_{1-x}N$  barrier need to be taken into consideration, will be further discussed below.

**Electrical model.** In the electrically-driven GaN:Eu QW device, carriers are injected into the GaN:Eu QW active region from the barriers. Our analysis is similar to the current injection efficiency analysis in a typical QW without the presence of RE elements<sup>54,55</sup>. The presence of  $Eu^{+3}$  ions modifies these rate equations to account for coupling with  $Eu^{+3}$  ions and complexes.

Previous experimental work on QW devices have shown that the carriers injected into the QW can escape to the barrier due to the high thermionic emission energy<sup>56</sup>. The thermionic-related carrier escape process needs to be accounted in the determination of IQE of electrically-driven QW based devices. In addition, the non-radiative and spontaneous radiative recombination process of carriers in the GaN host and  $Al_xGa_{1-x}N$  barreirs are also taken into consideration in the electrically-driven GaN:Eu QW.

The carrier rate equations both in the barrier ( $N_B$ ) and GaN:Eu QW active region ( $N_{OW}$ ) are given by:

$$\frac{\mathrm{dN}_{\mathrm{B}}}{\mathrm{dt}} = \frac{\mathrm{I}_{\mathrm{tot}}}{q \, V_{\mathrm{B}}} + \frac{\mathrm{N}_{\mathrm{QW}}}{\tau_{\mathrm{e}}} \frac{V_{\mathrm{QW}}}{V_{\mathrm{B}}} - \mathrm{N}_{\mathrm{B}} \left( \frac{1}{\tau_{\mathrm{B}}} + \frac{1}{\tau_{\mathrm{bw}}} \right) \tag{1}$$

$$\frac{dN_{QW}}{dt} = \frac{N_B}{\tau_{bw}} \frac{V_B}{V_{QW}} + \frac{N_c}{\tau_{diss}} \frac{V_{Eu}}{V_{QW}} - N_{QW} \left( \frac{1}{\tau_{nr}} + \frac{1}{\tau_{sp}} + \frac{1}{\tau_e} + \frac{1}{\tau_{c\_cap}} \right)$$
(2)

where, the  $V_{B_{c}}V_{QW}$ ,  $V_{Eu}$  are the volumes of the barrier, GaN:Eu QW and Eu-doped region of the GaN:Eu QW respectively. The  $I_{tot}$  is the total injected current in the barriers which is assumed to be equal to the total injected current into the device,  $\tau_{e}$  is the carrier thermionic escape time form the GaN:Eu QW active region to the barriers,  $\tau_{B}$  is the carrier lifetime in the barrier described by the non-radiative and spontaneous radiative processes in the barrier, and  $\tau_{bw}$  is the barrier-well lifetime<sup>54,57</sup>. The radiative and non-radiative carrier processes in the GaN host are described by the  $\tau_{sp}$  and  $\tau_{nr}$  respectively. In general, the non-radiative and spontaneous radiative recombination rates in the GaN host and  $Al_xGa_{1-x}N$  barriers are functions of the carrier concentrations in the QW and barrier, the bimolecular recombination coefficient B, Shockley-Hall-Read (SHR) constant A, and Auger coefficient C. More details regarding the non-radiative and spontaneous radiative region to the GaN host and  $Al_xGa_{1-x}N$  barriers, as well as the thermionic escape from GaN:Eu QW active region to the  $Al_xGa_{1-x}N$  barriers can be found in refs<sup>54,55,57-60</sup>.

The rate equations of complexes (N<sub>c</sub>) and excited Eu<sup>+3</sup> ions (N<sub>Eu</sub>) in the GaN:Eu QW active region are:

$$\frac{dN_c}{dt} = N_{QW}C_{c\_cap}\left(N_{traps} - N_c\right)\frac{V_{QW}}{V_{Eu}} + N_{Eu}C_{bt}\left(N_{traps} - N_c\right) - N_c\left(C_{tr}\left(N - N_{Eu}\right) + \frac{1}{\tau_{diss}} + \frac{1}{\tau_{c\_heat}}\right)$$

$$(3)$$

$$\frac{\mathrm{dN}_{\mathrm{Eu}}}{\mathrm{dt}} = \mathrm{N_cC_{tr}}(\mathrm{N} - \mathrm{N_{Eu}}) - \mathrm{N_{Eu}}\left(\mathrm{C_{bt}}\left(\mathrm{N_{traps}} - \mathrm{N_c}\right) + \frac{1}{\tau_{\mathrm{rad}}} + \frac{1}{\tau_{\mathrm{Eu\_heat}}}\right)$$
(4)

where, the N and N<sub>traps</sub> are the concentrations of Eu<sup>+3</sup> ions and traps in the GaN:Eu QW active region, respectively. The parameters  $C_{c\_cap}$ ,  $C_{bt}$  and  $C_{tr}$  are defined as the capture, back-transfer and transfer coefficients in cm<sup>3</sup>/s respectively. For the rate equations (3) and (4), a general capture, back-transfer and transfer rate can be defined as:

$$C_{c\_cap}\left(N_{traps} - N_{c}\right) = \frac{1}{\tau_{cap0}} \left(1 - \frac{N_{c}}{N_{traps}}\right) = \frac{1}{\tau_{c\_cap}}$$
(5)

$$C_{bt} \left( N_{traps} - N_c \right) = \frac{1}{\tau_{bt0}} \left( 1 - \frac{N_c}{N_{traps}} \right) = \frac{1}{\tau_{bt}}$$
(6)

$$C_{tr}(N - N_{Eu}) = \frac{1}{\tau_{tr0}} \left( 1 - \frac{N_{Eu}}{N} \right) = \frac{1}{\tau_{tr}}$$
(7)

Equations (5)–(7) account for saturation in the excited Eu<sup>+3</sup> concentration as well as in the concentration of formed complexes, when substituted in the rate equations (3) and (4). The subscript 0 denotes the relative capture, transfer and back-transfer rate and the term in the parenthesis denotes the degree of the respective excitation of Eu<sup>+3</sup> ion and the complex concentration. Thus, the terms of  $1/\tau_{c_ccap}$   $1/\tau_{tr}$  and  $1/\tau_{bt}$  can be viewed respectively as the general capture transfer and back-transfer rates of the system.

The injection efficiency of GaN:Eu QW active region is the ratio of the current arising from the radiative and non-radiative de-excitation of  $Eu^{+3}$  ions to the total current injected into the GaN:Eu QW system  $I_{tot}$  and can be expressed as:

$$\eta_{\rm inj\_electrical} = \frac{I_{\rm Eu}}{I_{\rm tot}},\tag{8}$$

where, the  $I_{Eu}$  represents the total recombination current arising from the radiative and non-radiative de-excitation of the  $Eu^{+3}$  ion and is defined as:

$$I_{Eu} = \frac{N_{Eu} q V_{Eu}}{\tau},\tag{9}$$

with

$$\frac{1}{\tau} = \frac{1}{\tau_{\rm rad}} + \frac{1}{\tau_{\rm Eu\_heat}},\tag{10}$$

where, the q is the electron charge.

Solving the system of equations (1)-(4) under steady state condition the current injection efficiency of the electrical model is obtained:

$$\eta_{\rm inj\_electrical} = \left[ \left(1 + \frac{\tau_{\rm bw}}{\left(\frac{1}{\tau_{\rm B}} + \frac{1}{\tau_{\rm bw}}\right)^{-1}}\right) \left(-\frac{\tau \ \tau_{\rm tr}}{\tau_{\rm Eu} \tau_{\rm diss}} + \frac{\tau \ \tau_{\rm c\_cap}}{\left(\frac{1}{\tau_{\rm nr}} + \frac{1}{\tau_{\rm sp}} + \frac{1}{\tau_{\rm c\_cap}}\right)^{-1}} \left(\frac{\tau_{\rm tr}}{\tau_{\rm Eu} \ \tau_{\rm comp}} - \frac{1}{\tau_{\rm bt}}\right) \right] - \frac{\tau \ \tau_{\rm c\_cap}}{\tau_{\rm e}} \left(\frac{\tau_{\rm tr}}{\tau_{\rm Eu} \ \tau_{\rm comp}} - \frac{1}{\tau_{\rm bt}}\right)^{-1}, \tag{11}$$

where, the  $1/\tau_{Eu}$  and  $1/\tau_{comp}$  are rates related to  $Eu^{+3}$  and complex:

$$\frac{1}{\tau_{\rm Eu}} = \frac{1}{\tau_{\rm rad}} + \frac{1}{\tau_{\rm Eu\_heat}} + \frac{1}{\tau_{\rm bt}},$$
(12)

$$\frac{1}{\tau_{\rm comp}} = \frac{1}{\tau_{\rm tr}} + \frac{1}{\tau_{\rm c\_heat}} + \frac{1}{\tau_{\rm diss}},\tag{13}$$

the internal quantum efficiency ( $\eta_{IOE \ electrical}$ ) for the electrical model is given by:

$$\eta_{\text{IQE\_electrical}} = \eta_{\text{inj\_electrical}} \cdot \eta_{\text{rad}}, \qquad (14)$$

where, the  $\eta_{rad}$  is the radiative efficiency of the Eu<sup>+3</sup> ions defined as the ratio of radiative to both radiative and non-radiative de-excitation of Eu<sup>+3</sup> ions:

$$\eta_{\rm rad} = \frac{N_{\rm Eu} / \tau_{\rm rad}}{N_{\rm Eu} / \tau} = \frac{\frac{1}{\tau_{\rm rad}}}{\frac{1}{\tau_{\rm rad}} + \frac{1}{\tau_{\rm Eu\_heat}}}.$$
(15)

**Optical model.** For the case of optically-pumped GaN:Eu QW, the thermionic emission rate from the GaN:Eu QW active region to the  $Al_xGa_{1-x}N$  barrier is neglected. In optically-pumped GaN:Eu QW, the excitation of the GaN host is resonant, and the generated carriers do not possess excess energy to escape the QW<sup>61-64</sup>. For the same reason, the  $Al_xGa_{1-x}N$  barriers are not excited and hence the non-radiative and radiative process of carriers in the barriers can be neglected.

In the optically-pumped GaN:Eu QW, the assumption that the GaN:Eu QW active region is excited resonantly above the bandgap with a photon flux  $\varphi$ , results in a rate equation of carriers in the GaN:Eu QW active region (N<sub>QW</sub>) of:

$$\frac{\mathrm{dN}_{\mathrm{QW}}}{\mathrm{dt}} = \alpha \varphi + \frac{\mathrm{N}_{\mathrm{c}}}{\tau_{\mathrm{diss}}} \frac{V_{\mathrm{Eu}}}{V_{\mathrm{QW}}} - \mathrm{N}_{\mathrm{QW}} \left( \frac{1}{\tau_{\mathrm{nr}}} + \frac{1}{\tau_{\mathrm{sp}}} + \frac{1}{\tau_{\mathrm{c-cap}}} \right),\tag{16}$$

where, the  $\alpha$  is the absorption coefficient of GaN and the  $\varphi$  is the photon flux of the excitation. The first term of the right part of equation(16) can be viewed as the corresponding current I<sub>tot</sub> arising from the creation of carriers due to absorption of the incident photon flux and is equal to:

$$a \ \varphi = \frac{I_{tot}}{q \ V_{QW}}. \tag{17}$$

The rate equations of complexes ( $N_c$ ) and excited Eu<sup>+3</sup> ions ( $N_{Eu}$ ) in the GaN:Eu QW active region are same as in the case of electrically-driven GaN:Eu QW and are given from equations (3)-(4). The injection efficiency for the optical model is defined as:

$$\eta_{inj\_optical} = \frac{I_{Eu}}{I_{tot}},$$
(18)

where, the  $I_{Eu}$  is defined from equation(9).

Solving the system of equations (3), (4) and (16) under steady state condition, the injection efficiency for the optical model is obtained:

#### **Electrical Injection Model**





**Figure 2.** Flow charts of electrical and optical current injection efficiency models. The blue boxes indicate the different levels of barrier, GaN:Eu QW, complex and Eu<sup>+3</sup> ion. Each level includes its own related processes. The levels are connected via the 'forward mechanisms' (black arrows), and via the 'recycling mechanisms' (red arrows).



The internal quantum efficiency for the optical model is given from equation(14) with the respective injection efficiency.

**Comparison between optical and electrical model.** The analysis of the current injection efficiency model indicates fundamental differences in the excitation path of  $Eu^{+3}$  ion in the GaN:Eu QW active region for the optically-pumped and electrically-driven GaN:Eu QW. In Fig. 2 a flow chart depicts the related mechanisms and phenomena along the excitation path of  $Eu^{+3}$  ion in the GaN:Eu QW for both models.

More specifically, the presence of the barrier level in the electrical model results in transport phenomena of the carriers. The effect of barrier-well lifetime which depends on the mobility of the carriers and the temperature T strongly influences the injection efficiency in the active region in a similar way as in the case of a QW without the presence of RE elements<sup>55,60</sup>. Additionally, recombination mechanisms (monomolecular, bimolecular and Auger recombination) also exist in the barrier. Further, the barrier opens an extra path for the carriers through the recycling mechanisms (red arrows in Fig. 2), increasing the probability of carrier deviation from the Eu<sup>+3</sup> excitation path. The thermionic escape from QW to the barrier, which is proportional to the concentration of carriers (N<sub>QW</sub>), becomes stronger with increasing the current density<sup>54,58,59</sup>. The transport phenomena and thermionic process limit the injection efficiency and internal quantum efficiency in the electrically-driven GaN:Eu QW device as opposed to optically-pumped GaN:Eu QW in which these phenomena do not exist.

#### **Simulation Results**

This section presents how the parameters such as SHR constant A, capture time  $\tau_{cap0}$ , transfer time  $\tau_{tr0}$ , back-transfer time  $\tau_{bt0}$ , dissociation time  $\tau_{diss}$ , and Eu<sup>+3</sup> radiative lifetime  $\tau_{rad}$ , affect the injection efficiency of electrically driven and optical-pumped Eu-doped GaN QW active region. The QW and barrier parameters used for the simulations, such as the values of effective masses and mobilities, can be found in reference<sup>57</sup>. The bimolecular recombination coefficient B and Auger coefficient C are fixed to  $10^{-11}$  cm<sup>3</sup>/s and  $10^{-32}$  cm<sup>6</sup>/s respectively<sup>57</sup>. Note that the A, B and C coefficients, which describe the radiative and non-radiative processes in the GaN host and Al<sub>x</sub>Ga<sub>1-x</sub>N barriers, are assumed to be the same for the barriers and the well. The Al composition was set at x = 10% for the Al<sub>x</sub>Ga<sub>1-x</sub>N barriers. Table 1 presents the parameters used in the numerical calculation of the injection efficiency for the GaN:Eu QW active region. References<sup>65,66</sup> were used as a starting point for the relative times between the GaN host, traps-complexes and Eu<sup>+3</sup> ions.

In our analysis, the injection efficiency  $(\eta_{inj\_optical}, \eta_{inj\_electrical})$  is plotted with the excited Eu<sup>+3</sup> concentration  $(N_{Eu})$  versus the photon flux  $(\varphi)$  - optical model - and input current density (J) - electrical model - (Figs 3–5). As shown in Figs 3–5, the injection efficiency of the Eu-doped GaN QW active region exhibits the droop

Parameters	Study I	Study II	Study III	Study IV	Study V
A $(10^7 s^{-1})$	0.1-1	1	1	1	1
$\tau_{cap0} (10^{-7} s)$	10	0.1-10	10	10	10
$\tau_{tr0}(10^{-7}s)$	360	360	3.6-360	360	360
$\tau_{diss}(10^{-3}s)$	1	1	1	0.0001-1	1
$\tau_{bt0}(10^{-6}s)$	200	200	200	2-200000	200
$\tau_{c\_heat}(10^{-3}s)$	1	1	1	1	1
$\tau_{Eu\_heat}(10^{-3}s)$	1	1	1	1	1
$\tau_{rad}(10^{-6}s)$	400	400	400	400	30-400
N (cm <sup>-3</sup> )	$1 \cdot 10^{19}$	$1 \cdot 10^{19}$	1.1019	$1.10^{19}$	1.1019
N <sub>traps</sub> (cm <sup>-3</sup> )	$1 \cdot 10^{19}$	$1 \cdot 10^{19}$	1.1019	$1.10^{19}$	1.1019
$L_{QW}, L_{Eu}, L_{B} (nm)$	2.5, 2.5, 5	2.5, 2.5, 5	2.5, 2.5, 5	2.5, 2.5, 5	2.5, 2.5, 5

**Table 1.** Parameters used for the numerical calculations of the current injection efficiency models. Study of individual parameters associated with the  $Eu^{+3}$  excitation path.



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**Figure 3.** Effect of Shockley-Hall-Read constant A on injection efficiency and excited Eu<sup>+3</sup> ion concentration of GaN:Eu QW active region. (a) Injection efficiency and excited Eu<sup>+3</sup> ion concentration as a function of photon flux for optical model and (b) Injection efficiency and excited Eu<sup>+3</sup> ion concentration as a function of current density for electrical model. The  $\eta_{IQE}$  is defined as  $\eta_{IQE} = \eta_{inj} \cdot \eta_{rad}$  and follows the same trend as the  $\eta_{inj}$  of the optical and electrical model.



**Figure 4.** Effect of capture time  $\tau_{cap0}$  on injection efficiency and excited  $Eu^{+3}$  ion concentration of GaN:Eu QW active region. (a) Injection efficiency and excited  $Eu^{+3}$  ion concentration as a function of photon flux for optical model and (b) Injection efficiency and excited  $Eu^{+3}$  ion concentration as a function of current density for electrical model. The  $\eta_{IQE}$  is defined as  $\eta_{IQE} = \eta_{inj} \cdot \eta_{rad}$  and follows the same trend as the  $\eta_{inj}$  of the optical and electrical model.

characteristics. Since the excited  $Eu^{+3}$  concentration cannot exceed the maximum available  $Eu^{+3}$  concentration in the active region, the excited  $Eu^{+3}$  concentration increases with the photon flux and the current density. At a point where the excited  $Eu^{+3}$  concentration saturates due to the maximum available  $Eu^{+3}$  concentration in the active region, the subsequent increase of photon flux and current density leads to the droop in the injection



**Figure 5.** Effect of transfer time  $\tau_{tr0}$  on injection efficiency and excited  $Eu^{+3}$  ion concentration of GaN:Eu QW active region. (a) Injection efficiency and excited  $Eu^{+3}$  ion concentration as a function of photon flux for optical model and (b) Injection efficiency and excited  $Eu^{+3}$  ion concentration as a function of current density for electrical model. The  $\eta_{IQE}$  is defined as  $\eta_{IQE} = \eta_{inj}$ . $\eta_{rad}$  and follows the same trend as the  $\eta_{inj}$  of the optical and electrical model.

efficiency. The rate of this saturation and the droop in the injection efficiency depend on the values of the different parameters related to specific mechanisms in the excitation path.

**Study I: Effect of Shockley-Hall-Read constant.** The SHR constant A is related to the non-radiative process of monomolecular recombination which takes place through defects in the crystal lattice. SRH mechanism has been shown to be a critical process affecting the injection efficiency of light emitting diodes<sup>57</sup>. As shown in Fig. 3, at low photon fluxes and current densities, the injection efficiency is higher as the SRH constant is smaller. Such characteristic is expected, since lower values of SRH constant indicate lower non-radiative recombination rates of carriers in the active region and barrier. As a result, the injection efficiency in the GaN:Eu QW active region increases for optical and electrical model. Interestingly, it should be noted that the increase of the SHR constant A would lead to slower saturation of the excited Eu<sup>+3</sup> concentration as the photon flux and current density is increasing. This indicates that additional carriers are required through optical excitation in the optically-pumped device or electrical injection in the electrically-driven device to replace the carriers lost in the monomolecular non-radiative recombination process. Thus, higher photon fluxes and current densities are required to result in same Eu<sup>+3</sup> excitation as opposed to lower values of A.

**Study II: Effect of capture time.** The capture of carriers from traps with a rate  $1/\tau_{cap0}$  results in the creation of complexes. A general capture time  $\tau_{c_{cap0}}$  is given from equation(5) which is a function of the formed complexes (N<sub>c</sub>). Figure 4 shows the effect of capture time  $\tau_{cap0}$  both for optical and electrical model. Following the previous analysis, as the capture time decreases, the carriers are captured more efficiently from traps increasing the formation rate of complexes and consequently the excited Eu<sup>+3</sup> concentration. This efficient capture of carriers from traps increases the injection efficiency and decrease the required amount of photon fluxes and current densities. This is observed as a shift towards lower photon fluxes and current densities of the excited Eu<sup>+3</sup> concentration and injection efficiency for both models. For the optical model, the higher injection efficiency occurs for the lower capture time of  $\tau_{cap0} = 10^{-7}$ s where the injection efficiency drops from  $\eta_{inj\_optical} = 21\%$  to  $\eta_{inj\_optical} = 0.2\%$ . In contrast, for the electrical model it drops from  $\eta_{inj\_electrical} = 9\%$  to  $\eta_{inj\_electrical} = 0.01\%$ .

**Study III: Effect of transfer time.** The transfer time defines the rate at which complexes de-excite by releasing energy to a nearby Eu<sup>+3</sup> ion. As shown in Fig. 5, the injection efficiency increases as the transfer time  $\tau_{tr0}$  decreases, which is a result of the faster de-excitation of the complexes. Equation(4) indicates that the de-excitation rate of complexes,  $1/\tau_{tr}$  is essentially the excitation rate of Eu<sup>+3</sup> ions. As a result, the higher excitation rates of Eu<sup>+3</sup> ions result in faster saturation of excited Eu<sup>+3</sup> concentration under steady state conditions. This is observed as a shift toward lower photon fluxes ( $\phi$ ) and current densities (J) of the excited Eu<sup>+3</sup> concentration. For the given range of photon flux and current density, the values of  $\tau_{tr0} = 36 \times 10^{-6}$  s and  $\tau_{tr0} = 36 \times 10^{-7}$  s result in saturation of excited Eu<sup>+3</sup> concentration close to the value of Eu<sup>+3</sup> ion concentration in the active region (N =  $1 \times 10^{19}$  cm<sup>-3</sup>), while the value of  $\tau_{tr0} = 36 \times 10^{-5}$  s results in saturation N<sub>Eu</sub> =  $\sim 4 \times 10^{18}$  cm<sup>-3</sup> which is almost 40% of the total concentration of Eu<sup>+3</sup> ion in the GaN:Eu QW active region.

**Study IV: Effect of complex dissociation rate and energy back-transfer rate.** As stated before, the complexes can dissociate, releasing the captured electrons and holes into the QW with a rate of  $1/\tau_{diss}$ . Similarly, the excited Eu<sup>+3</sup> ions can de-excite with a rate  $1/\tau_{bi0}$  by releasing energy which results in the formation of complexes. Both dissociation time and back-transfer time are related to processes which can be considered as recycling mechanisms: in the case of dissociation process, the resulted electrons and holes can be re-captured from traps to form complexes, while in the back-transfer process the formed complexes can result to the excitation of Eu<sup>+3</sup> ions. For this study, five different values of back-transfer and dissociation rates are selected for a given current density J = 0.87 A/cm<sup>2</sup> and photon flux of  $\varphi = 4 \times 10^{19} \text{ cm}^{-2} \text{s}^{-1}$ .



**Figure 6.** Injection efficiency and excited Eu<sup>+3</sup> ion concentration of GaN:Eu QW active region as a function of (a) back-transfer rate  $1/\tau_{bt0}$  and (b) dissociation rate  $1/\tau_{diss}$ . The  $\eta_{IQE}$  is defined as  $\eta_{IQE} = \eta_{inj} \cdot \eta_{rad}$  and follows the same trend as the  $\eta_{inj}$  of the optical and electrical model. The two models are compared for the same values of Eu<sup>+3</sup> excited ion concentration in the GaN:Eu QW active region.





As it is shown in Fig. 6, by increasing the dissociation rate, the injection efficiency and excited Eu<sup>+3</sup> concnetration drop significantly. More specifically, for the electrical model injection efficiency drops from  $\eta_{inj\_electrical} = 0.18\%$  to almost  $\eta_{inj\_electrical} = 0.001\%$ , while for the optical model drops from  $\eta_{inj\_optical} = 0.9\%$  to almost  $\eta_{inj\_optical} = 0.01\%$ . The changes in excited Eu<sup>+3</sup> concnetration are identical for the two models.

A droop in the injection efficiency and excited Eu<sup>+3</sup> concentration with the back-transfer rate is also observed for both models. More specifically, the droop starts when the back-transfer rate of  $1/\tau_{bt0} = 5 \times 10^4 \, s^{-1}$  becomes comparable with the transfer rate of complexes,  $1/\tau_{tr0} = 2.77 \times 10^4 \, s^{-1}$ . For back-transfer rates lower than  $1/\tau_{bt0} = 5 \times 10^4 \, s^{-1}$ , the injection efficiency and excited Eu<sup>+3</sup> concentration remain unaffected.

In addition, the changes in the injection efficiency and excited  $Eu^{+3}$  concnetration with the back-transfer rate, are smaller as compared to the changes with the complex dissociation rate. As it can be seen from Fig. 2, the level at which the dissociation process takes place is distant from the level of  $Eu^{+3}$  ion. Thus, the carriers resulted from the dissociation of complexes have higher probability to deviate from the  $Eu^{+3}$  excitation path reducing in that way the injection efficiency and the excited  $Eu^{+3}$  concentration in the GaN:Eu QW active region.

Study V: Effect of radiative lifetime of  $Eu^{+3}$  ion – Enhancement of radiative efficiency. The parameters presented in the previous sections affect the internal quantum efficiency of the system by altering only the injection efficiency in the active region. The internal quantum efficiency is calculated from equation(14) with a radiative efficiency fixed at  $\eta_{rad} = ~72\%$  and follows the same trend of the injection efficiency. The radiative lifetime ( $\tau_{rad}$ ) and the non-radiative time ( $\tau_{Eu\_heat}$ ) of  $Eu^{+3}$  ion determine the radiative efficiency of the GaN:Eu QW system. Lower radiative lifetime results in higher radiative efficiencies, assuming that the non-radiative lifetime of  $Eu^{+3}$  ion remains unchanged.

By reducing the radiative lifetime, the injection efficiency and excited  $Eu^{+3}$  concentration are significantly altered. The lower radiative lifetime indicates faster radiative de-excitation rate of excited  $Eu^{+3}$  ions, therefore,

higher injection efficiency can be achieved at a given photon flux and current density. This is clearly illustrated in Fig. 7. In addition, the resulted lower saturation values of excited  $Eu^{+3}$  ions, make the injection efficiency to be strongly altered at higher photon fluxes and current densities.

Attributing to the differences in complex interplays among the fundamental processes in the current injection process, the optical model exhibits higher injection efficiency as compared to the electrical model for the same values of excited Eu<sup>+3</sup> concentration. In particular, the reduction of radiative lifetime from  $\tau_{rad} = 400 \,\mu s$  to  $\tau_{rad} = 30 \,\mu s$ , changes the excited Eu<sup>+3</sup> concentration from  $N_{Eu} = 8.4 \times 10^{18} \, cm^{-3}$  to  $N_{Eu} = 3.25 \times 10^{18} \, cm^{-3}$  at a given  $\varphi = 4.7 \times 10^{18} \, cm^{-3}$ . Meanwhile, injection efficiency increases from  $\eta_{inj\_optical} = 0.62\%$  to  $\eta_{inj\_optical} = 2.4\%$  which is 3.8 times higher. A similar change in the excited Eu<sup>+3</sup> concentration occurs at J = 1 A/cm<sup>2</sup> for the electrical model while the injection efficiency increases from  $\eta_{inj\_electrical} = 0.12\%$  to  $\eta_{inj\_electrical} = 0.46\%$  which is almost 3.8 times higher, same change as in the optical model. The reduction of the radiative lifetime is essential for achieving higher injection efficiencies at higher photon fluxes and current densities, while at the same time the radiative efficiency of Eu<sup>+3</sup> ions is enhanced.

#### Comparison with experimentally reported data

In order to compare our work with experimentally reported values of GaN:Eu devices, the external quantum efficiency ( $\eta_{EQE}$ ) for a GaN:Eu QW LED with a square device area of 1000 × 1000 µm is calculated. The external quantum efficiency is the product of the extraction efficiency ( $\eta_{extr}$ ) and the internal quantum efficiency of the device. An extraction efficiency of  $\eta_{extr} = 44\%$  was used for our calculations, which is a typical value for GaN:Eu based device<sup>48</sup>. The details of each simulation are given in Table 2a and b. The numerical calculations for the external quantum efficiency are divided into two groups: Group A represents those which resulted in  $\eta_{EQE} > 1\%$  and Group B represents those which resulted in  $\eta_{EQE} < 1\%$ .

Figure 8(a) presents our numerical calculations and experimentally reported values with two different types of GaN:Eu based LED. A. Nishikawa and co-workers fabricated two GaN:Eu based LEDs with a 300 nm GaN:Eu active layer each, under different growth conditions<sup>44</sup>. They reported an external quantum efficiency of  $\eta_{EQE} = 0.6\%$  at an injected current of 0.5 mA which was found to reduce to  $\eta_{EQE} = 0.04\%$  at 20 mA. W. Zhu and co-workers fabricated a GaN:Eu based LED with an active layer of alternate GaN/GaN:Eu regions and they reported an external quantum efficiency of  $\eta_{EQE} = 4.6\%$  at an injected current of 1 mA which reduced to  $\eta_{EQE} = 0.9\%$  at 20 mA<sup>49</sup>. These values correspond to the highest reported external quantum efficiency up to date. The calculated EQE from the electrical current injection efficiency model, follows the same trend as the experimentally reported values.

In addition, both experimental studies revealed that higher injected current into the GaN:Eu device led to saturation in the EL spectra, which was attributed to the saturation of the excited Eu<sup>+3</sup> ions. Similar findings have also been reported elsewhere<sup>45,47</sup>. Our study is consistent with the experimental observations that increasing the injected current will eventually result in the saturation of the excited Eu<sup>+3</sup> concentration with a subsequent decrease in the injection efficiency and internal quantum efficiency the GaN:Eu QW active region.

#### Engineering the IQE of electrically-driven GaN:Eu QW

The increase of the injection efficiency, as well as its shift at higher input current densities is the desirable goal for highly efficient electrically-driven GaN:Eu based red light emitters. The numerical calculations from the current injection efficiency model, showed the pathway for high efficiency in the GaN:Eu QW system. By physically engineering the factors in the GaN:Eu QW system that affect injection efficiency ( $\eta_{inj}$ ) is thus critical for achieving high internal quantum efficiency ( $\eta_{IQE}$ ) in electrically-driven GaN:Eu QW based devices.

**Material quality.** The SHR constant A is related to defects present in the barrier and QW. Higher values of A reflect poor material quality which is detrimental for IQE of the device. The numerical calculation of IQE of the electrically-driven GaN:Eu QW device, showed an increase 62% in the IQE at 5 mA when the SHR constant A decreased by 98% (Fig. 8(b), Simulation I and Simulation II). High quality  $Al_xGa_{1-x}N$  alloy material with low defect concentration can be fabricated with advanced growth techniques, such as MOCVD by carefully adjusting the growth parameters<sup>67–70</sup>. This low defect concentration suppresses the SHR mechanisms and is expected to increase the IQE of GaN:Eu QW based device.

**Carrier capture process, back-transfer process and complex related processes.** The capture time of carries form traps, the lifetime parameters related to complexes and the back-transfer process, depend on the nature of those elements as well as on the interaction between them and with the host. This work quantitatively verified that a significant enhancement of the IQE of the electrically-driven GaN:Eu QW device can be achieved by changing these lifetimes. A decrease of 90% both in capture time and transfer time resulted in an IQE increase of ~33% at 5 mA and ~24% at 20 mA (Fig. 8(b), Simulation II and Simulation III). Recent studies have shown that through defect engineering such as co-doping with magnesium (Mg), and also through manipulation of growth conditions, can result in the enhancement of the IQE of GaN:Eu device<sup>44,49,50,52,53,71</sup>.

**Europium and trap concentration in the GaN host and thermionic emission process.** Another parameter that can be engineered is the number of available traps  $(N_{traps})$  and  $Eu^{+3}$  ion concentration (N). In this work, the effect of these two parameters is not presented. However, these values can be modified to increase the injection efficiency. More specifically, by increasing the amount of available traps  $(N_{traps})$ , the general capture rate according to equation(5) will be increased giving rise to the injection efficiency. Similarly, the simultaneous increase of  $Eu^{+3}$  ion concentration (N) will also increase the transfer rate according to equation(7). As a result, the injection efficiency will be increased at higher input current densities and photon fluxes.



**Figure 8.** (a) EQE calculated from the electrical current injection efficiency model of GaN:Eu QW device and experimentally reported values of GaN:Eu based LED. (b) Calculated IQE for the electrically-driven GaN:Eu QW device. The simulation parameters are shown in Table 2(a).


Parameters	Simulation I	Simulation II	Simulation III	Simulation IV				
(a)								
A (s <sup>-1</sup> )	0.5.108	106	106	106				
$\tau_{cap0}(s)$	10 <sup>-7</sup>	10 <sup>-7</sup>	10 <sup>-8</sup>	10 <sup>-8</sup>				
$\tau_{tr0}(s)$	36.10-6	36.10-6	36.10-7	36.10-7				
$\tau_{diss}, \tau_{bt0} (s)$	10 <sup>-3</sup> , 200.10 <sup>-6</sup>							
$\tau_{c\_heat}, \tau_{Eu\_heat} (s)$	10 <sup>-3</sup> , 10 <sup>-3</sup>							
$\tau_{rad}(s)$	200.10-6	200.10-6	200.10-6	100.10-6				
N (cm <sup>-3</sup> )	8.5·10 <sup>19</sup>	8.5·10 <sup>19</sup>	8.5·10 <sup>19</sup>	8.5·10 <sup>19</sup>				
N <sub>traps</sub> (cm <sup>-3</sup> )	8.5·10 <sup>19</sup>	8.5·10 <sup>19</sup>	8.5·10 <sup>19</sup>	8.5·10 <sup>19</sup>				
$L_{QW}$ , $L_{Eu}$ , $L_B$ (nm)	5, 5, 10	5, 5, 10	5, 5, 10	5, 5, 10				
(b)								
A (s <sup>-1</sup> )	106	106	106	106				
$\tau_{cap0}(s)$	10 <sup>-4</sup>	10 <sup>-6</sup>	10 <sup>-6</sup>	10 <sup>-6</sup>				
$\tau_{tr0}(s)$	36.10-6	36.10-6	36.10-4	36.10-6				
$\tau_{diss}, \tau_{bt0} (s)$	10 <sup>-3</sup> , 200.10 <sup>-6</sup>	10 <sup>-6</sup> , 200.10 <sup>-6</sup>	10 <sup>-3</sup> , 200.10 <sup>-6</sup>	10 <sup>-3</sup> , 200.10 <sup>-8</sup>				
$\tau_{c\_heat}, \tau_{Eu\_heat} (s)$	10 <sup>-3</sup> , 10 <sup>-3</sup>							
$\tau_{rad}(s)$	200.10-6	200.10-6	200.10-6	100.10-6				
N (cm <sup>-3</sup> )	8.5·10 <sup>19</sup>	8.5·10 <sup>19</sup>	8.5·10 <sup>19</sup>	8.5·10 <sup>19</sup>				
N <sub>traps</sub> (cm <sup>-3</sup> )	8.5·10 <sup>19</sup>	8.5·10 <sup>19</sup>	8.5·10 <sup>19</sup>	8.5·10 <sup>19</sup>				
$L_{QW}$ , $L_{Eu}$ , $L_B$ (nm)	5, 5, 10	5, 5, 10	5, 5, 10	5, 5, 10				

**Table 2.** (a) Simulations of external quantum efficiency (EQE) for a GaN:Eu QW device-high EQE. (b) Simulations of external quantum efficiency for a GaN:Eu QW device-low EQE.

In our study the Al composition of the barrier was set to 10%. Higher Al composition will increase the conduction and valence band offsets and will suppress the carrier thermionic escape the barrier<sup>57</sup>. Thus, engineering the barrier height for carriers is crucial for higher injection efficiencies in the GaN:Eu QW active region.

**Radiative lifetime of Eu<sup>+3</sup> ion.** The radiative efficiency of Eu<sup>+3</sup> ion can also be modified through the engineering of radiative lifetime of Eu<sup>+3</sup> ions. Our numerical calculations of IQE of the electrically-driven GaN:Eu QW device, showed that changing the radiative lifetime from  $\tau_{rad} = 200 \ \mu s$  to  $\tau_{rad} = 100 \ \mu s$  results in an increase

of ~144% of the IQE at 20 mA of the GaN:Eu QW device (Fig. 8(b), Simulation III and Simulation IV). It has been experimentally demonstrated that by utilizing surface-plasmon (SP) in GaN-based QW can dramatically increase the radiative efficiency of the system<sup>72-75</sup>. The photon density states near the SP frequency ( $\varpi_{sp}$ ) are increased from Purcell factor. For the case of GaN:Eu QW system, by carefully engineering the deposited materials used as SP, the SP frequency can be adjusted to coincides with the frequency of the emitted photons of Eu<sup>+3</sup> ions. This approach will increase the radiative efficiency and consequently the internal quantum efficiency of the GaN:Eu based devices.

#### Summary

In summary, we developed a physically intuitive current injection efficiency model for optically-pumped and electrically-driven GaN:Eu QW and we demonstrated the pathway for enhancing the internal quantum efficiency ( $\eta_{IQE}$ ) of a GaN:Eu QW system. It was shown that the saturation of excited Eu<sup>+3</sup> concentration with the photon flux and input current density, is the main cause of the current injection efficiency droop in the GaN:Eu QW active region. Through the manipulation of the characteristic times along the excitation path of Eu<sup>+3</sup> ion, the injection efficiency ( $\eta_{inj}$ ) and internal quantum efficiency ( $\eta_{IQE}$ ) of GaN:Eu QW system can be significantly enhanced. In addition, the discrepancy between the efficiencies of optically-pumped and electrically-driven GaN:Eu QW is explained in the framework of the current injection efficiency models. Our findings through the analysis within the current injection efficiency model also clarify the necessary means towards the practical realization of highly efficient red light GaN:Eu QW LED.

#### References

- Keller, S. et al. Gallium nitride based high power heterojunction field effect transistors: Process development and present status at UCSB. IEEE Transactions on Elect. Dev. 48, 3 (2001).
- 2. Oka, T. & Nozawa, N. AlGaN/GaN recessed MIS-gate HFET with high-threshold-voltage normally-off operation for power electronics applications. *IEEE Elect. Dev. Lett.* 29, 7 (2008).
- 3. Tong, H. *et al.* Thermoelectric properties of lattice-matched AlInN alloy grown by metal organic chemical vapor deposition. *Appl. Phys. Lett.* **97**, 112105 (2010).
- 4. Yamaguchi, S., Izaki, R., Kaiwa, N., Sugimura, S. & Yamamoto, A. Thermoelectric devices using and thin films prepared by reactive radiofrequency Sputtering. *Appl. Phys. Lett.* 84, 5344 (2004).
- 5. Dahal, R., Pantha, B., Li, J., Lin, J. Y. & Jiang, H. X. InGaN/GaN multiple quantum well solar cells with long operating wavelengths. *Appl. Phys. Lett.* **94**, 063505 (2009).
- 6. Kirste, R. et al. Electronic biosensors based on III-nitride semiconductors. Annual Rev. Analytical Chem. 8, 149-169 (2015).

7. Nakamura, S., Senoh, M., Iwasa, N. & Nagahama, S. High-brightness InGaN blue, green and yellow light-emitting diodes with quantum well structures. *Jap. J. Appl. Phys.* **34**(Part 2), Number 7A (1995).

- 8. Nakamura, S. et al. InGaN-based multi-quantum-well structure laser diodes. Jap. J. Appl. Phys. 35(Part 2), Number 1B (1996).
  - . Masui, H., Nakamura, S., DenBaars, S. P. & Mishra, U. K. Nonpolar and semipolar III-nitride light-emitting diodes: Achievements and challenges. *IEEE Transactions. on Elect. Dev.* 57, 1 (2010).
- 10. Tansu, N. et al. III-Nitride Photonics. IEEE Photonics Journal 2, 241-248 (2010).
- 11. Steigerwald, D. A. et al. Illumination with, solid state lighting technology. IEEE J. on Select. Top. in Quant. Elect. 8, 2 (2002).
- 12. Krames, M. R. Status and future of high power light-emitting diodes for solid-state lighting. J. of Disp. Tech. 3, 2 (2007).
- 13. Krames, M. R. Status and future prospects for visible-spectrum light-emitting diodes. Digest of tech. papers 47, 39-41 (2016).
- 14. Tsao, J. Y. et al. Toward smart and ultra-efficient solid-state lighting. Adv. Optical Mater. 2, 809-836 (2014).
- 15. Belyaev, K. G. et al. Phase separation in  $In_xGa_{1-x}N$  (0.10 < x < 0.40). Phys. Status Solidi C 10, 527–531 (2013).
- 16. McCluskey, M. D. et al. Phase separation in InGaN multiple quantum wells. Appl. Phys. Lett. 72, 1730–1732 (1998).
- Takeuchi, T. et al. Quantum-confined stark effect due to piezoelectric fields in GaInN strained quantum wells. Jap. J. Appl. Phys. 36(Part 2), Number 4A (1997).
- 18. Damilano, B. & Gil, B. Yellow-red emission from (Ga,In)N heterostructures. J. Phys. D: Appl. Phys. 48, 403001 (2015).
- 19. Xu, G. et al. Investigation of Large Stark Shifts in InGaN/GaN Multiple Quantum Wells. J. Appl. Phys. 113, 033104 (2013).
- Zhao, H. P. et al. Approaches for high internal quantum efficiency green InGaN light-emitting diodes with large overlap quantum qells. Optics Express 19, A991–A1007 (2011).
- Zhao, H. P. Growths of staggered InGaN quantum wells light-emitting diodes emitting at 520-525 nm employing graded growthtemperature profile. *Appl. Phys. Lett.* 95, 061104 (2009).
- Zhao, H. P., Arif, R. A., Ee, Y. K. & Tansu, N. Self-consistent analysis of strain-compensated InGaN-AlGaN quantum wells for lasers and light emitting diodes. *IEEE J. Quantum Electron.* 45, 66–78 (2009).
- Zhang, J. & Tansu, N. Improvement in spontaneous emission rates for InGaN quantum wells on ternary InGaN substrate for lightermitting, diodes. J. Appl. Phys. 110, 113110 (2011).
- 24. Daubler, J. et al. Long wavelength emitting GaInN quantum wells on metamorphic GaInN buffer layers with enlarged in-plane lattice parameter. Appl. Phys. Lett. 105, 111111 (2014).
- Ohkawa, K., Watanabe, T., Sakamoto, M., Hirako, A. & Deura, M. 740-nm emission from InGaN based LEDs on c-plane sapphire substrates by MOVPE. J. Crys. Growth 343, 13–16 (2012).
- Hwang, J., Hashimoto, R., Saito, S. & Nunoue, S. Development of InGaN-based red LED grown on (0001) polar surface. Appl. Phys. Express 7, 071003 (2014).
- Kawaguchi, Y. *et al.* Semipolar (2021) single-quantum-well red light-emitting diodes with a low forward voltage. *Jap. J. Appl. Phys.* 52, 08JC08 (2013).
- Zhao, H. P., Liu, G. Y. & Tansu, N. Analysis of InGaN-delta-InN quantum wells for light-emitting diodes. Appl. Phys. Lett. 97, 131114 (2010).
- 29. Tan, C. K., Borovac, D., Sun, W. & Tansu, N. InGaN/Dilute-As GaNAs interface quantum well for red emitters. *Sci. Reports* 6, 19271 (2016).
- Favennec, P. N., L'Haridon, H., Salvi, M., Moutonnet, D. & Le Guillou, Y. Luminescence of erbium implanted in various semiconductors: IV, III-V and II-VI materials. *Electron. Lett.* 25, 718–719 (1989).
- 31. Kenyon, A. J. Recent developments in rare-earth doped materials for optoelectronics. *Progress in Quantum Elect.* 26, 225–284 (2002).
- Hömmerich, U. et al. Photoluminescence studies of rare earth (Er, Eu, Tm) in situ doped GaN. Mat. Sc. Eng. B 105, 91–96 (2003).
   Lozykowski, H. J., Jadwisienczak, W. M., Han, J. & Brown, I. G. Luminescence properties of GaN and Al0.14Ga0.86N/GaN
- superlattice doped with europium. Appl. Phys. Lett. 77, 767 (2000).
  34. Sawahata, J., Bang, H., Seo, J. & Akimoto, K. Optical processes of red emission from Eu doped GaN. Sc. Tech. Adv. Mat. 6, 644–648 (2005).

- 35. Wakamatsu, R. et al. Luminescence Properties of Eu-Doped GaN Grown on GaN Substrate. Jap. J. Appl. Phys. 52, 08JM03 (2013).
- M. de Boer, W. D. A. *et al.* Optical excitation and external photoluminescence quantum efficiency of Eu<sup>3+</sup> in GaN. *Sci. Rep.* 4, 5235 (2014).
- Wakamatsu, R., Timmerman, D., Lee, D., Koizumi, A. & Fujiwara, Y. Afterglow of Eu related emission in Eu-doped gallium nitride grown by organometallic vapor phase epitaxy. J. Appl. Phys. 116, 043515 (2014).
- Heikenfeld, J., Garter, M., Lee, D. S., Birkhahn, R. & Steckl, A. J. Red light emission by photoluminescence and electroluminescence from Eu-doped GaN. Appl. Phys. Lett. 75, 1189 (1999).
- 39. Kim, J. H. & Holloway, P. H. Room-temperature photoluminescence and electroluminescence properties of sputter grown gallium nitride doped with europium. J. Appl. Phys. 95, 4787 (2004).
- 40. Park, J. H. & Steckl, A. J. Laser action in Eu-doped GaN thin-film cavity at room temperature. *Appl. Phys. Lett.* **85**, 4588 (2004).
- Nishikawa, A., Kawasaki, T., Furukawa, N., Terai, Y. & Fujiwara, Y. Room-temperature red emission from a p-type/europiumdoped/n-type gallium nitride light-emitting diode under current injection. *Appl. Phys. Express.* 2, 071004 (2009).
- Nishikawa, A., Kawasaki, T., Furukawa, N., Terai, Y. & Fujiwara, Y. Electroluminescence properties of Eu-doped GaN-based red light-emitting diode by OMVPE. *Phys. Stat. Sol. A* 207, 1397–1399 (2010).
- Nishikawa, A., Furukawa, N., Kawasaki, T., Terai, Y. & Fujiwara, Y. Improved luminescence properties of Eu-doped GaN lightemitting diodes grown by atmospheric-pressure organometallic vapor phase epitaxy. *Appl. Phys. Lett.* 97, 051113 (2010).
- 44. Nishikawa, A., Furukawa, N., Kawasaki, T., Terai, Y. & Fujiwara, Y. Room-temperature red emission from light-emitting diodes with Eu-doped GaN grown by organometallic vapor phase epitaxy. *Optical Mat.* **33**, 1071–1074 (2011).
- 45. Sekiguchi, H. et al. Red light-emitting diodes with site-selective Eu-doped GaN active layer. Jap. J. Appl. Phys. 52, 08JH01 (2013).
- Ishii, M., Koizumi, A. & Fujiwara, Y. Nanoscale determinant to brighten up GaN: Eu red light-emitting diode: Local potential of Eudefect complexes. J. App. Phys. 117, 155307 (2015).
- Arai, T. et al. Enhanced excitation efficiency of Eu ions in Eu-doped GaN/AlGaN multiple quantum well structures grown by organometallic vapor phase epitaxy. J. Luminescence 158, 70–74 (2015).
- Inaba, T. et al. Substantial enhancement of red emission intensity by embedding Eu-doped GaN into a microcavity. AIP Adv. 6, 045105 (2016).
- Zhu, W. et al. High-power Eu-doped GaN red LED based on a multilayer structure grown at lower temperatures by organometallic vapor phase epitaxy. MRS Adv. 67, 159–164 (2017).
- 50. Fujiwara, Y. & Dierolf, V. Present understanding of Eu luminescent centers in Eu-doped GaN grown by organometallic vapor phase epitaxy. *Jap. J. Appl. Phys.* 53, 05FA13 (2014).
- Woodward, N. *et al.* Excitation of Eu<sup>3+</sup> in gallium nitride epitaxial layers: Majority versus trap defect center. *Appl. Phys. Lett.* 98, 011102 (2011).
- 52. Mitchell, B. *et al.* The role of donor-acceptor pairs in the excitation of Eu-ions in GaN:Eu epitaxial layers. *J. Appl. Phys.* **115**, 204501 (2014).
- Masago, A., Fukushima, T., Sato, K. & Katayama-Yoshida, H. Efficient luminescent center by codoping (Eu,Mg,O) into GaN. Appl. Phys. Express. 7, 071005 (2014).
- 54. Tansu, N. & Mawst, L. J. Current injection efficiency of 1300-nm InGaAsN quantum-well lasers. J. Appl. Phys. 97, 054502 (2005).
- Nagarajan, R., Ishikawa, M., Fukushima, T., Geels, R. S. & Bowers, J. E. High speed quantum-well lasers and carrier transport effects. IEEE J. Quantum. Electron. 28, 1990–2008 (1992).
- 56. Xu, L. F. *et al.* Experimental evidence of the impact of nitrogen on carrier capture and escape times in InGaAsN/GaAs single quantum well. *IEEE Photonics J.* **4**, 2262–2271 (2012).
- 57. Zhao, H. P., Liu, G. Y., Zhang, J., Arif, R. A. & Tansu, N. Analysis of internal quantum efficiency and current injection efficiency in nitride light-emitting diodes. J. Disp. Technol. 9, 212–225 (2013).
- Schneider, H. and v. Klitzing, K., Thermionic emission and gaussian transport of holes in a GaAs/Al<sub>x</sub>Ga<sub>1-x</sub>As multiple-quantumwell structure. *Phys. Rev. B* 38, 6160 (1988).
- Tansu, N. & Mawst, L. J. The Role of Hole-Leakage in 1300-nm InGaAsN Quantum Well Lasers. Appl. Phys. Lett. 82, 1500–1502 (2003).
- 60. Taylor, G. W. & Jin, S. Revisions to "Transport solution for SCH QW laser diodes". IEEE J. Quantum. Electron. 34, 1886–1889 (1998).
- 61. Gardner, N. F. et al. Blue-emitting InGaN-GaN double-heterostructure light-emitting diodes reaching maximum quantum efficiency above 200A/cm<sup>2</sup>. Appl. Phys. Lett. **91**, 243506 (2007).
- 62. Shen, Y. C. et al. Auger recombination in InGaN measured by photoluminescence. Appl. Phys. Lett. 91, 141101 (2007).
- 63. Kim, M. et al. Origin of efficiency droop in GaN-based light-emitting diodes. Appl. Phys. Lett. 91, 183507 (2007).
- 64. Xie, J. et al. On the efficiency droop in InGaN multiple quantum well blue light emitting diodes and its reduction with p doped quantum well barriers. *Appl. Phys. Lett.* **93**, 121107 (2008).
- Lee, C. W., Everitt, H. O., Lee, D. S., Steckl, A. J. & Zavada, J. M. Temperature dependence of energy transfer mechanisms in Eudoped GaN. J. Appl. Phys. 95, 7717 (2004).
- Wang, J., Koizumi, A., Fujiwara, Y. & Jadwisienczak, W. M. Study of defects in GaN *in situ* doped with Eu<sup>3+</sup> ion grown by OMVPE. J. Elect. Mat. 45, 2001–2007 (2016).
- Amano, H., Sawaki, N., Akasaki, I. & Toyoda, Y. Metalorganic vapor phase epitaxial growth of a high quality GaN film using an AlN buffer layer. Appl. Phys. Lett. 48, 353 (1986).
- 68. Hashimoto, T., Wu, F., Speck, J. S. & Nakamura, S. A GaN bulk crystal with improved structural quality grown by the ammonothermal method. *Nature Mater.* **6**, 568 (2007).
- 69. Nakamura, S. In Situ Monitoring of GaN growth using interference effects. Jap. J. Appl. Phys. 30, 10A (1991).
- Ee, Y. K. et al. Metalorganic vapor phase epitaxy of III-nitride light-emitting diodes on nano-patterned AGOG sapphire substrate by abbreviated growth mode. IEEE J. Sel. Top. Quantum Electron. 15, 1066–1072 (2009).
- 71. Zavada, J. M. Impurity co-doping of gallium nitride materials for enhanced light emission. ECS Trans. 61, 65–70 (2014).
- 72. Okamoto, K. et al. Surface plasmon-enhanced light emitters based on InGaN quantum wells. Nature Mater. 3, 601-605 (2004).
- 73. Okamoto, K. *et al.* Surface plasmon enhanced spontaneous emission rate of InGaN/GaN quantum wells probed by time-resolved photoluminescence spectroscopy. *Appl. Phys. Lett.* **87**, 071102 (2005).
- 74. Neogi, A. *et al.* Enhancement of spontaneous emission in a quantum well by resonant surface plasmon coupling. *Phys. Rev. B* 66, 153305 (2002).
- 75. Zhao, H. P., Zhang, J., Liu, G. Y. & Tansu, N. Surface plasmon dispersion engineering via double-metallic Au/Ag layers for III-nitride based light-emitting diodes. *Appl. Phys. Lett.* **98**, 151115 (2011).

#### Acknowledgements

This work was supported by US National Science Foundation (ECCS 1408051 and DMR 1505122), and the Daniel E. '39 and Patricia M. Smith Endowed Chair Professorship Fund.

### **Author Contributions**

I.E.F., C.K.T., and N.T. contributed to the discussions, concept development, theoretical analysis, analysis of the results, and writing of the manuscript. V.D., and Y.F. contributed to the technical discussions, and analysis of the results. N.T. supervised the studies performed in the manuscript.

#### Additional Information

**Competing Interests:** The authors declare that they have no competing interests.

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